

Development of a Manufacturing Process for High-Precision Cu EOS Targets

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OVERVIEW

This document describes the development of a manufacturing process and the production of Cu EOS targets. The development of a manufacturing process for these targets required a great deal of research, because the specifications for the targets required a level of precision an order of magnitude beyond Target Fabrication's capabilities at the time. Strict limitations on the dimensions of the components and the interfaces between them required research efforts to develop bonding and deposition processes consistent with a manufacturing plan with a dimensional precision on the order of 0.1 µm. Several months into this effort, the specifications for the targets were relaxed slightly as a result of discussions between the Target Fabrication Group and the physicists. The level of precision required for these targets remained an order of magnitude beyond previous capabilities, but the changes made it possible to manufacture targets to the specifications. The development efforts and manufacturing processes described in this document successfully produced a complete Cu EOS target that satisfied all of the fabrication and metrology specifications.

TARGET SPECIFICATIONS

An exploded view of a Cu EOS target appears in Figure 1. Several different target designs were originally planned for shot series, including 16 kJ shots on the Omega laser, 23 MBar shots on NIF, and 60 MBar shots on NIF. Each of these shot series was originally planned to include targets with Cu samples and Al samples. Each of these targets was similar to the 23 MBar Cu NIF target illustrated in the figure. The required dimensions of each of the EOS targets changed several times over the course of this effort, but the final design for the 23 MBar Cu NIF target included a baseplate of 1100 Al of thickness 62 μ m. A Cu sample of thickness 43 μ m is attached to one face of the baseplate, and a 102 μ m pre-heat shield composed of 6.25 at% ICH is attached to the other face. A 32.5 μ m CH ablator is bonded to the drive side of the pre-heat shield. Each of these components has a diameter of approximately 4 mm. Quartz windows of thickness 100 μ m are bonded to the VISAR side of the baseplate and sample.

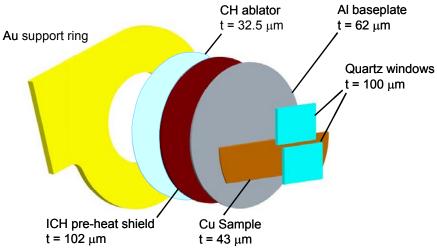


Figure 1. Exploded view of the 23 MBar Cu NIF EOS target

The original specifications for the targets represented a level of precision far beyond the capabilities of the Target Fabrication Group. One of the most important requirements for the targets was the interface between the Al baseplate and the Cu sample. The original specification for the targets required that this interface have a thickness no larger than 0.5 μ m. In addition, the original specification required that the surface of the baseplate have no short-wavelength features with a peak-to-valley height larger than 0.1 μ m. The required thickness uniformities of several of the components were originally 0.1 μ m over the central 500 μ m diameter region, and the absolute thickness and thickness uniformity of several of the components were required to be metrologized with an accuracy on the order of 0.1 μ m.

The original specifications for the targets governed the majority of the development effort. The specifications for the targets were later relaxed slightly as a result of discussions between the Target Fabrication Group and the physicists. One of the most significant changes was that the new specifications allowed a step of up to 1 μ m in the Al baseplate at the edge of the Cu sample, provided this step was measured with an accuracy of 0.2 μ m. In the final specifications, the required thickness uniformities of the various layers of the target ranged from $\pm 0.1~\mu$ m to $\pm 0.3~\mu$ m, and they had to be measured with this same level of accuracy. A summary of the thickness uniformity specifications appears in Figure 2. The bond layers for the CH and ICH components had to be less than 3 μ m in thickness. Quartz windows of thickness 100 μ m were bonded to the VISAR side of the baseplate and sample, and the bonds for the windows had to be less than 5 μ m in thickness. Extensive collaboration between the physicists and the Target Fabrication Group was required to obtain specifications that could be realized in the manufacturing process and satisfied the physics requirements for the targets.

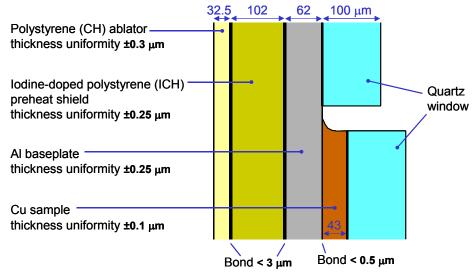


Figure 2. Summary of the specifications for the target that drove the development of the manufacturing process

A complete list of all of the specifications and acceptance criteria for the final target appears in Appendix A.

PRELIMINARY INVESTIGATIONS OF MANUFACTURING PROCESSES

Designing a manufacturing process to meet the physics requirements proved to be a challenging task. The two issues that were far beyond current capabilities at the commencement

of this effort were the fixturing of components on the machine tool to obtain the required dimensional accuracy and metrology, and the ability to meet the interface requirements between the Cu sample and the Al baseplate.

Workpiece Fixturing

The standard practice used by the Target Fabrication Group to manufacture a component with a diameter of a few mm is to attach the component to a standard holder, like that shown in Figure 3. These holders are semi-kinematic mounts that can be placed onto and removed from the diamond turning machines with a repeatability of approximately 0.25 μ m. These holders are used to produce target components with sub- μ m accuracy, which is usually more than adequate.



Figure 3. Standard Al holder used to mount workpieces on the DTM

Unfortunately, these standard holders do not have adequate precision to achieve thickness uniformity and metrology on the order of $0.1~\mu m$. Some simple tests were performed to estimate the repeatability of these holders. The tests indicated that when inserting and removing these holders from the diamond turning machine, angular errors of up to 0.014 deg can occur. In order to meet the requirements for the targets, a different type of workpiece fixturing with greater precision was required.

Several different methods for fixturing the workpieces were considered. The method selected uses a "narrow land" vacuum chuck to hold a workpiece of diameter 100 mm, as shown in Figure 4. A vacuum chuck with lands smaller than 50-75 μ m is known to be able to hold flat workpieces with a repeatability of 25 nm (1 μ in) when used properly. The workpiece diameter of 100 mm makes the parts easy to handle, so they can be properly inserted and removed from the vacuum chuck with the required accuracy. The large diameter of the workpiece also has the advantage that any axial runout of the workpiece due to a seating error on the vacuum chuck will correspond to a small angular error, which leads to machined parts with improved thickness uniformity. The selected configuration has a number of other advantages. The large workpiece contains sufficient area to measure the seating of the workpiece on the vacuum chuck using either an LVDT or a capacitance probe, which is required to obtain the required metrology of the targets. And finally, a large number of 4 mm diameter targets could be fabricated from a single 100 mm disk, which appeared to be a significant advantage when this plan was being developed and the shot schedule called for dozens of EOS targets.

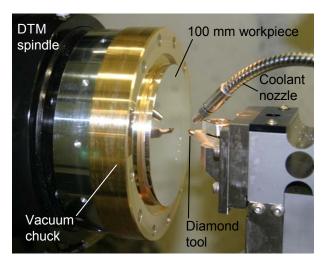
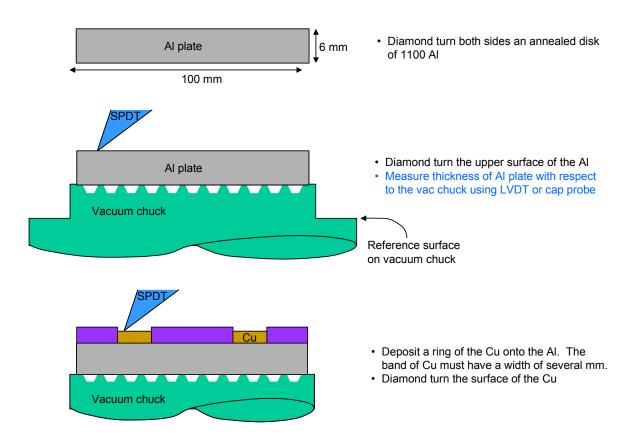


Figure 4. Machining of a 100 mm diameter workpiece on a "narrow land" vacuum chuck

Outline of Manufacturing Plan

The basic outline of the manufacturing plan for these targets is illustrated in the following figure. A number of details were added to this plan since its inception and will be discussed later, but this basic premise was used when designing the manufacturing process for the targets. A 100 mm diameter disk of 1100 Al as the primary workpiece. This Al disk is machined to create the baseplate of the target, and the sample, pre-heat shield, and ablator are added during the manufacturing process.



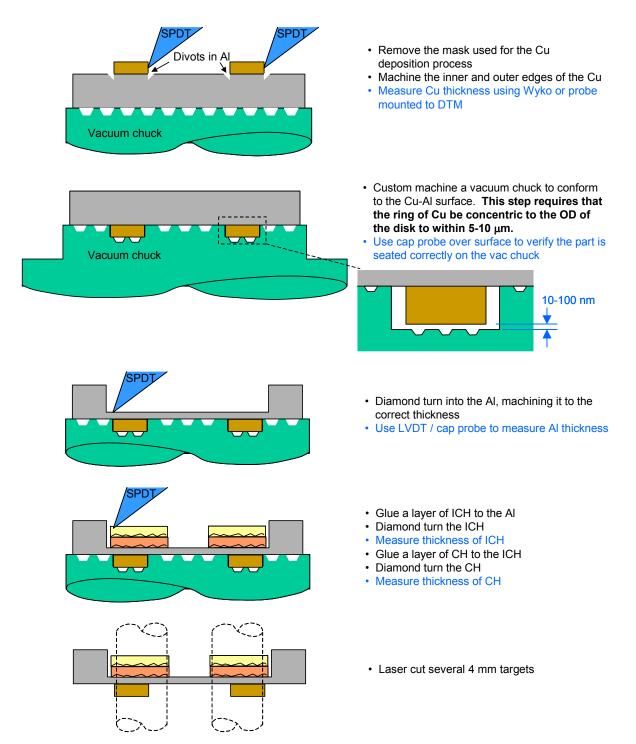


Figure 5. Schematic illustration of the basic outline for the manufacturing process for the EOS targets

At each step of the this manufacturing process, the fixturing of the workpiece is crucial to obtaining the required precision. Each time the 100 mm disk is placed on the vacuum chuck, its seating on the chuck is carefully checked by measuring the axial runout using a capacitance probe. This step is not illustrated in Figure 5, but it is crucial to meeting the specifications for the targets. Each of the steps in this process was carefully designed to be consistent with the

required overall precision of the targets. One of the challenging steps in this process was depositing the Cu onto the Al baseplate with adequate adhesion without damaging the 100 mm Al disk.

Interface between the Cu sample and the Al baseplate

When this target fabrication effort began, the required interface between the Cu sample and the Al baseplate of the targets could not be manufactured using the available standard fabrication practices. An adhesive bond created with the Target Fabrication Groups's standard bonding methods using epoxy or cyanoacrylate was known to have a thickness on the order of a few μm , so a different means of attaching the Cu to the Al had to be devised. A research effort was required to develop a method of joining the Cu to the Al in a manner consistent with the subsequent precision manufacturing steps. The difficulty in this task was to bond the Cu and the Al with an interface thickness less than 0.5 μm using a method that maintained all of the reference surfaces and workpiece form and accuracy, so the subsequent steps in the manufacturing process produced targets with the required 0.1 μm thickness uniformity and allowed the targets to be metrologized with an accuracy of 0.1 μm . Developing such a method for bonding the Cu to the Al required several months of effort.

Several different bonding methods were considered for bonding the Cu sample and Al baseplate of the target. Initial efforts focused on using a low-viscosity hydroxide solution, which formed a joint by attaching to the oxide resident on the Cu and the Al and bonding them together. This method was shown to produce bondlines of thickness 1 µm or smaller. However, difficulties in controlling the amount of adhesive dispensed resulted in excess solution being applied that completely covered the workpiece components and did not maintain the required environment for producing precision targets. Therefore, this method initially showed good potential for producing sub-µm bonds, but it was not pursued beyond the preliminary investigations in favor of other methods.

Several different methods of directly depositing Cu onto Al were investigated in detail. Depositing sub-µm layers of Cu onto an Al substrate using sputtering, electroplating, and e-beam evaporation were all standard practices for which the capability existed at LLNL. Each of these methods had several advantages and disadvantages, and it was not initially known which of them was best suited for fabricating the EOS targets. Therefore, several tests were performed to investigate each of the processes. To reduce the time and cost of the experiments, investigations were performed by applying Cu to disks of 1100 Al of thickness 6 mm and diameter 25 mm. These smaller Al disks could be prepared fairly quickly. The following paragraphs summarize the most important findings of these experiments.

E-beam evaporation

An experiment was performed in which $82~\mu m$ of Cu was deposited on a 25~mm diameter Al disk using an E-beam evaporation process. The roughness of the as-deposited Cu was R_t =80 nm. The Cu did not experience any problems with adhesion while being diamond turned. The interface between the Cu and the Al appeared to have met the specifications for the targets. Unfortunately, SEM pictures of this interface were not made available to the Target Fabrication Group. This process appeared to be a good method of obtaining adequate adhesion with an acceptable interface. One of the major drawbacks to this method is that it is performed at a temperature of several hundred $^{\circ}$ C. The temperature is too high for a Kapton mask, so a metal

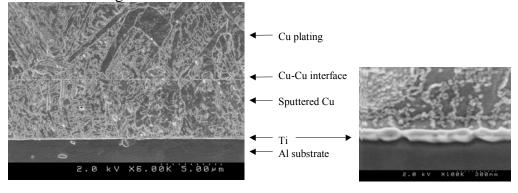
hardmask must be used, which introduces the problem of aligning the hardmask with the reference surfaces on the workpiece to within 5 to $10 \mu m$.

Zincated electroplating

Several attempts were made to electroplate Cu onto 25 mm Al disks. Because of the oxide layer on Al, Cu will not adhere to the Al unless the Al surface is prepared properly. One quick and easy method of preconditioning the Al is to zincate the surface, which involves chemically removing the oxide and then electroplating Zn onto the newly cleaned surface. Cu can then be electroplated onto the Zn. Unfortunately, the zincating process creates a surface roughness of Rt=120 to 600 nm, depending on the process parameters. In addition, up to 200 nm of Al can be removed during the process. Therefore, if a section of the Al disk is masked with tape, then a step of up to 200 nm will be formed between the zincated surface and the masked Al surface. Because of the roughness created by this process, it could not be used to produce targets that met the specifications.

Sputter-seeded electroplating

Several trials were performed by sputtering a few µm of Cu onto the Al disk to provide a conduction path for an electroplating process. After masking a portion of a 25 mm Al disk with Kapton tape, the oxide was removed with an ion mill, and 20 nm of Ti were sputtered followed by 3 µm of Cu. The sputtering was done in B231. Then an additional 30 µm of Cu were electroplated onto the sputtered Cu in LLNL's Metal Finishing Shop. This Cu did not experience any problems with adhesion while being diamond turned. One advantage of this process is that Kapton tape can be applied directly to the Al surface to create an effective mask. A sample that was prepared using this method was sectioned and examined with an SEM, as shown in Figure 6. These images were taken from a sample that was prepared several months after this preliminary study, as described in Appendix B, but it was prepared in a very similar manner and should be representative of the condition of the samples that were prepared during these preliminary tests. The images reveal that the interface between the Cu and the Al meets the specifications for the targets.



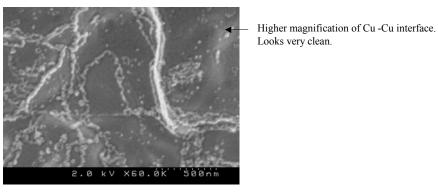


Figure 6. SEM images of a sample prepared by sputter-seeded electroplating

Sputtering Cu to the required thickness

A trial was also performed in which 30 µm of Cu was sputtered directly onto a 25 mm Al disk in B231. This sputtering process took several days to complete. The sputtered Cu did not experience any problems with adhesion while being diamond turned. However, there was some concern that the stresses in sputtered Cu could be very large and could eventually lead to delamination. Delamination was not actually observed in this test, but the concerns that were voiced caused this potential method to be abandoned in favor of other methods. Note that this project did not find any direct evidence to indicate whether or not sputtering Cu to the required thickness would be a successful process, and it is possible that it could have worked.

These preliminary investigations indicated that E-beam evaporation and sputter-seeded electroplating were good candidates for depositing the Cu onto the 100 mm Al disk. Therefore, both of these processes were attempted on full-scale 100 mm Al disks.

100 mm Al DISKS MACHINED WITH MINERAL OIL

First Attempt at Sputter-Seeded Electroplating (5/26/04)

The first attempt to fabricate a batch of targets used a sputter-seeded electroplating process to deposit a 4 mm wide ring of Cu on the 100 mm Al disk. The 100 mm Al disk was prepared by diamond turning both sides using mineral oil as the cutting fluid. After machining the disk, it was cleaned with an alcohol rinse. A layer of Kapton tape was then laid onto one surface of the disk. Two circular score marks were made through the Kapton using the diamond tool on the diamond turning machine, and the resulting ring of Kapton tape was peeled up to create a 4 mm wide circular band of exposed Al, as shown in Figure 7. Because this scoring was done directly on the diamond turning machine, the circular band of exposed Al was concentric to the outer diameter of the disk, which was critical for the future step in the manufacturing process where the 100 mm disk must be inserted into a vacuum chuck with the Cu face seated against the vacuum chuck.

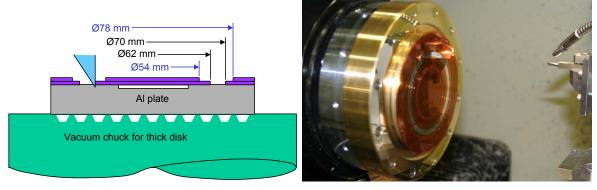
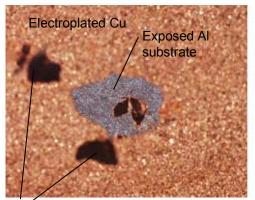


Figure 7. Schematic illustration and photograph of the scoring and removal of the Kapton ring

The disk was then sputtered in B235 with 10 nm of Ti followed by 3 μ m of Cu. Note that the sputtering chamber in B235 was not equipped with an ion mill, so this process differed from the initial sputter-seeded electroplating tests performed on the 25 mm diameter Al sample. Additional Cu was then electroplated onto the sputtered Cu. The Cu that had been electroplated onto the part was visibly blistered and appeared to have delaminated from the 100 mm Al disk. A scalpel was used to pick through the blistered Cu. Photographs of the area in which flakes of Cu were removed appear in Figure 8.





Flakes of Cu broken from surface

Back surface of a flake of Cu

Figure 8. Flakes of the blistered electroplated Cu and the underlying Al substrate

The underside of the flake of material pried from the surface was silver in color, not copper. Therefore, it appeared that the sputtered Ti was well bonded to the sputtered Cu, and the delamination occurred at the interface between the Ti and the Al. The left side of Figure 9 depicts a portion of the Al disk that was protected by Kapton during sputtering and electroplating. This image reveals that residual traces of the adhesive from the Kapton tape remained on the surface of the Al after the Kapton was removed. The right side of Figure 9 depicts the Al substrate in the area in which the flake of Cu was removed. This image reveals a texture and some contaminants on the Al. The features on this surface could have been related to the Kapton residue that was left after the 4 mm wide ring of Kapton tape was removed from this area.

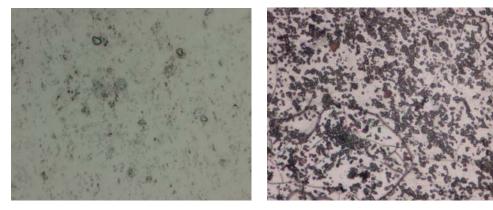


Figure 9. Microscope images at 500× of a portion of the Al disk that was protected by Kapton during sputtering and electroplating (left) and the Al beneath the delaminated Cu (right)

First Attempt at E-Beam Evaporation of Cu (6/10/04)

Another attempt was made to deposit Cu onto a 100 mm Al disk using an E-beam evaporation process. This process was more complicated than the sputtering process, because it required a precision hardmask. This hardmask had to withstand a temperature of several hundred °C, and it had to locate the deposited Cu on the 100 mm Al disk very precisely. The ring of deposited Cu had to be concentric to the OD of the 100 mm disk to within approximately 5 to 10 µm. This requirement was very important, because the disk was later flipped over into a vacuum chuck with the Cu side against the vacuum chuck, and the OD of the disk was used as a reference surface during this operation. If the Cu was not concentric to the OD of the disk, then the Cu would have been damaged during this step, or the Al disk could have been scratched, which would have eliminated a crucial precision reference surface. The stainless steel precision hardmask made for the E-beam evaporation of Cu onto the 100 mm Al disk is shown in Figure 10.

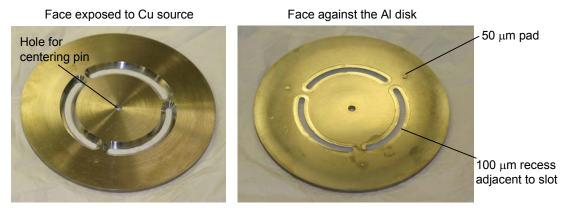
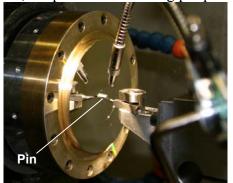
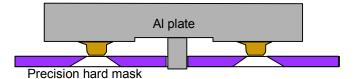


Figure 10. Precision hardmask used for E-beam evaporation of Cu onto the 100 mm Al disk

Prior to doing the deposition of the Cu, the hardmask was placed in the deposition chamber in the configuration shown in the right side of Figure 10. The Al disk was placed on top of the hardmask, where it rested on three pads of height 50 μ m. These pads supported the disk above the surface of the hardmask and ensured that the Al disk was scratched only at the location of the pads. The hardmask contained three tapered grooves of width 4 mm, through which the Cu was deposited. A recess of depth 100 μ m adjacent to each groove prevented the newly deposited Cu from sticking to the hardmask. The required concentricity of the deposited

Cu with respect to the OD of the Al disk was obtained by using a precision centering pin that had been press-fit into the center of the Al disk and diamond turned to be concentric to the OD of the disk, as shown in Figure 11. This pin was only a few µm smaller than the hole machined in the center of the hardmask. Thus, when the disk was assembled into the mask as shown in the figure, the precision centering pin provided the required concentricity.

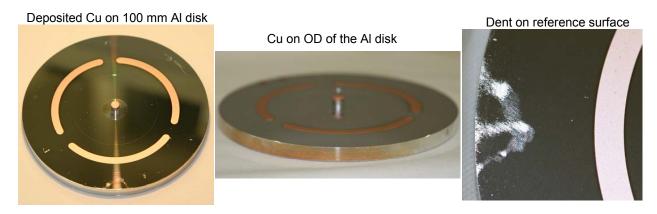




Schematic illustration of the placement of the Al disk on top of the hardmask for E-beam evaporation of Cu onto the 100 mm Al disk

Figure 11. Machining of the precision locating pin and installation of the disk onto the hardmask

Upon receiving the disk back from the E-beam evaporation process, several flaws were apparent on the disk. Some Cu had been deposited on the OD of the Al disk. Although the Cu deposited on this reference surface was less than 1 μ m in thickness and was not itself a problem, it indicated that the process had not been executed as planned. A much larger flaw was also evident on the surface of the Al disk onto which the Cu had been deposited. This precision reference surface had been damaged during the deposition and contained a dent that raised the surface by several μ m in a localized area. Therefore, one of the required precision reference surfaces was lost, and the disk could not be used to make acceptable targets.



It was later discovered that an error occurred while installing the 100 mm disk in the hardmask. The locating pin on the disk caught in the hole in the hardmask, so the disk did not seat properly onto the mask during the deposition. This problem is illustrated in Figure 12. The temperatures reached in the deposition process could easily have caused the 100 mm Al disk to warp due to creep.

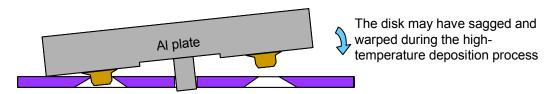


Figure 12. Misalignment of the 100 mm disk and the hardmask

To determine the quality of the Cu edge created by the E-beam evaporation process with this hardmask, the edge of the deposited Cu was measured with the Wyko NT 8000 optical profiling system. A profile trace taken through the measured dataset appears in Figure 13. The profile trace reveals that the deposited Cu had a height of approximately 53 μ m. The sidewall of the deposited Cu was not perfectly vertical, and there was a "shadow" where the thickness of the Cu tapered toward zero. This "shadow" had a height of approximately 1 μ m at a radial distance of 100 μ m from the edge of the Cu, and the "shadow" disappeared at a distance of approximately 1 μ m from the edge of the Cu.

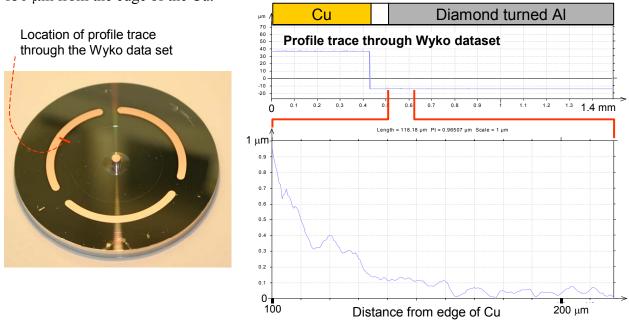


Figure 13. Wyko measurement of the shadow at the edge of the deposited Cu

Second Attempt at Sputter-Seeded Electroplating (6/23/04)

The problem that was believed to have occurred with the first attempt at sputter-seeded electroplating of the Cu was poor adhesion of the sputtered Ti caused by residual adhesive on the surface of the Al from the Kapton tape. In an attempt to fix the problem and obtain better adhesion, another 100 mm Al disk was processed. This disk was prepared by diamond turning both sides using mineral oil as the cutting fluid. After machining the disk, it was cleaned with an alcohol rinse, and a layer of Kapton tape was then applied to one surface. As with the previous disk, two circular score marks were made through the Kapton using the diamond tool on the diamond turning machine, and the resulting ring of Kapton tape was then peeled off to create a 4 mm wide circular band of exposed Al. In an attempt to remove any residual adhesive from the Kapton tape, the 4 mm wide circular band of exposed Al was washed with acetone. To perform the acetone rinse, the disk was mounted on the spindle of the diamond turning machine and

rotated at several hundred rpm. A squirt bottle of acetone was then used to apply a stream of acetone to the exposed Al for several seconds, and a squirt bottle of isopropyl alcohol was used to rinse off the acetone. The disk was then sputtered in B235 with 10 nm of Ti followed by 3 μ m of Cu. Several hundred μ m of Cu were then electroplated onto the sputtered layer of Cu.

Plated Cu

Al disk

Electroplating mask

Figure 14. Electroplated disk



Figure 15. Delaminating Cu

Figure 14 depicts the Kapton-masked 100 mm Al disk inside the hardmask used for the electroplating process. This photograph was taken just after removing the disk and the electroplating mask from the plating bath. After removing the disk from this hardmask, the Kapton was removed, and the disk was placed back on the vacuum chuck on the diamond turning machine. The Cu was then diamond turned to reduce its thickness. At the time, this disk was intended for 16 kJ Omega EOS targets, which required a Cu sample of thickness 25 μ m. As the Cu was being machined to the correct thickness, it began to delaminate and peel away from the Al disk, as shown in Figure 15. At a thickness of 27 μ m, a portion of the Cu completely fell off of the Al disk. The backside of the Cu strip was silver in appearance, which indicated that once again, the sputtered Ti had delaminated from the Al substrate.

Second Attempt at E-Beam Evaporation of Cu (7/20/04)

One of the problems that occurred during the first attempt to deposit Cu using E-beam evaporation was that the locating pin on the disk caught in the hole in the hardmask, so the disk did not seat properly onto the mask during the deposition. Another attempt was made to deposit Cu onto a 100 mm Al disk using E-beam evaporation, but this time the precision locating pin was machined differently. Rather than machining a simple cylindrical pin, the pin was machined in a barrel shape. The curved edges of the pin allowed the disk to be correctly inserted onto the hardmask in the E-beam evaporation chamber.

By this point in time, the 16 kJ Omega EOS shots had been cancelled. Therefore, this disk was intended to be used to make 23 MBar NIF EOS targets, which required a Cu sample of thickness 43 μ m. It was well known that material deposited by E-beam evaporation can have a surface roughness of several hundred nm, and the upper and lower surfaces of the deposited material can be crowned or "shadowed" as shown in Figure 13. Therefore, excess material must always be deposited so it can be diamond turned to the desired thickness. Therefore, the Target Fabrication Group requested that 50 to 60 μ m of Cu be deposited using the E-beam evaporation process.

The disk received from the E-beam evaporation process did not have sufficient Cu. The thickness of the deposited Cu ranged from 38 to 44 μm . The disk was mounted back on the diamond turning machine, and the Cu was machined to a thickness of 41.5 μm . This machining process removed Cu from a portion of the disk, but in some areas, the Cu was not touched by the cutting tool, as shown in Figure 16.

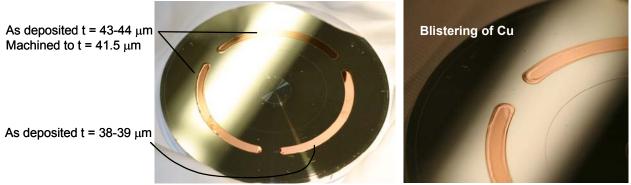


Figure 16. Cu deposited by E-beam evaporation after machining it to a thickness of 41.5 μm

The Cu deposited in this process was not adequately adhered to the Al disk. The right side of Figure 16 reveals that the Cu began to blister near the edges and delaminate from the Al. The 4 mm wide band of deposited Cu was inspected with the Wyko NT8000 optical profiling system. A profile trace made through the dataset revealed that the diamond turned surface of the Cu had a profile of approximately 0.5 µm over the 4 mm width, as shown in Figure 17. This profile was caused by Cu that blistered from the Al after the machining process.

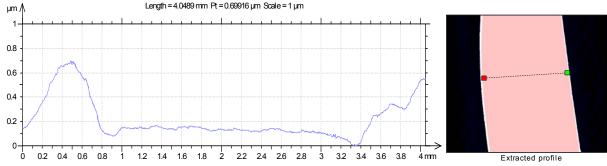


Figure 17. Profile trace across the 4 mm wide band of Cu

CLEANING TESTS

After attempts to deposit Cu onto the Al disk using both sputter-seeded electroplating and E-beam evaporation suffered from delamination of the Cu, it was apparent that the processes needed to be changed to obtain better adhesion. Therefore, attempts to produce targets were put on hold, and a research effort was performed to determine how adhesion of the Cu to the Al could be improved.

It was hypothesized that the delamination of the Cu could have been related to inadequate cleaning of the Al disk. The research effort used samples of diameter 25 mm to perform tests on different cleaning processes. Examples of the 25 mm disks are depicted in Figure 18. The left side of the figure depicts a 25 mm disk in the sputtering chamber in B231 immediately after it had been sputtered with 5 μ m of Cu. The right side of the figure depicts a 25 mm disk that was coated with Cu using E-beam evaporation. The Kapton mask had been peeled off of this disk.

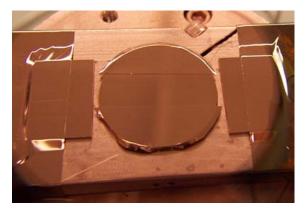




Figure 18. Disks of diameter 25 mm used for cleaning tests

The tests performed on these 25 mm disks yielded a great deal of valuable information on the procedures exercised prior to and during the Cu deposition operations. Complete details of the tests are available in Appendix B. A summary of some of the important observations is given below.

- Mineral spirits can be used as a cutting fluid to machine diamond turned surfaces on Al 1100. Because mineral spirits is much more volatile than mineral oil, it evaporates more quickly and is more easily removed from the surface. Therefore, it is recommended that all machining be performed using mineral spirits as the cutting fluid.
- Pentane is an excellent solvent for cleaning mineral spirits off of diamond turned surfaces. One cleaning method that led to adequate adhesion of the Cu to the Al was to clean the diamond turned surface by rinsing it with pentane, followed by acetone, followed by clean ethanol. This sequence should be repeated three times. This cleaning method does not have any noticeable adverse effect on the surface finish of diamond turned Al. After machining a sample using mineral spirits and cleaning it three times with pentane, acetone, and ethanol, a SIMS analysis did not identify any peaks representative of mineral spirits, pentane, acetone, or ethanol.
- Pentane should not be stored in polyethylene bottles. The pentane will become contaminated and deposit crystals on the Al that are not easily removed. A microscope image of these crystals appears in Figure 19. Pentane should be stored in a glass bottle and can be applied using a syringe with a syringe filter or with a glass pipette.

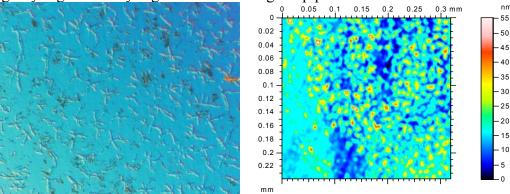


Figure 19. Microscope image (500×) and Wyko dataset (PSI 20×) of debris left by pentane that had been stored in a polyethylene bottle

- Prior to being sputtered, parts should not be stored in polyethylene boxes, because the outgassing of the plastic can contaminate the Al. Instead, parts should be stored in Fluoroware trays, which are commonly used as wafer shipping trays and do not outgas significantly.
- Alcohol and acetone should not be used in the vicinity of Kapton tape. They can dissolve the adhesive on the tape and transport a white residue over the surface of the parts. This white residue can be very detrimental to the adhesion of deposited Cu.
- Bob Dively Liquid Masking Film can be applied as a barrier between the Al and the Kapton tape. This masking film can be peeled up, and SIMS analyses indicated that it does not leave any significant residue on the surface.
- Performing a SIMS analysis does produce any noticeable detrimental effect on surface finish
- A 30 second swirl in dilute Oakite solution has a small detrimental effect on surface finish.
 In one test on a diamond turned Al surface, the Oakite cleaning process increased S_t from 30 nm to 40 nm.
- Cu should not be directly E-beam evaporated onto Kapton tape. (The parameters used for this test were not made available to the Target Fabrication Group). At the test conditions used for this single test, the Kapton warped and peeled off of the Al, leaving a significant amount of adhesive residue behind. This residue was easily removed with acetone.

100 mm Al DISKS MACHINED WITH MINERAL SPIRITS

The cleaning tests identified some procedural changes that should have promoted adhesion of the deposited Cu to the Al. The most important discovery was that mineral spirits makes an excellent cutting fluid and can be removed by rinsing the part with pentane, acetone, and clean ethanol. This cleaning process appeared to lead to excellent adhesion of the deposited Cu to the Al. This information was incorporated into resumed production attempts on 100 mm Al disks.

Test Disk for Sputter-Seeded Electroplating (9/30/04)

While the cleaning tests were taking place, several 100 mm diameter Al disks were diamond turned such that the two faces were parallel to within 100 nm. Because more than one month had elapsed since a 100 mm disk was run through the sputtering and electroplating processes, a non-precision test disk was run through the process with the intent of working out any bugs in the different steps.

This test disk was machined using mineral spirits as the cutting fluid. It was then rinsed with pentane, acetone, and ethanol. The cleaning process was repeated three times. The disk was then masked with Kapton tape. However, rather than applying the tape directly to the Al surface, a layer of Bob Dively Liquid Masking Film was first applied to the disk, as illustrated in Figure 20.

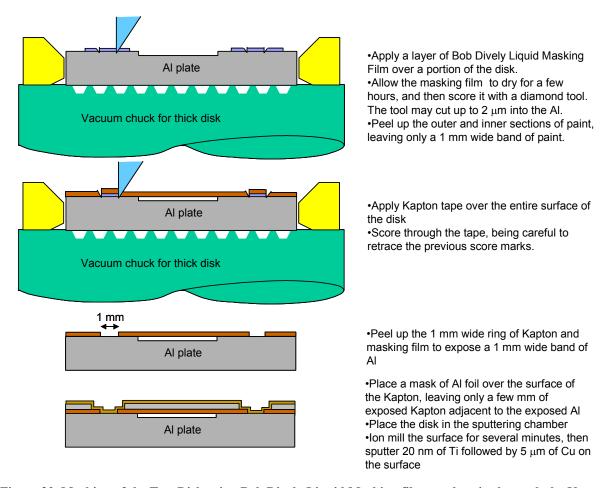


Figure 20. Masking of the Test Disk using Bob Dively Liquid Masking film as a barrier beneath the Kapton

The disk was then placed in the sputtering chamber in B231, ion milled for several minutes, and sputtered with 20 nm of Ti followed by 5.4 µm of Cu. Figure 21 depicts the setup of the Test Disk in the sputtering chamber. The foil masks that have been placed over the Kapton prevent the ion milling process from ablating too much of the Kapton. The right side of the figure depicts the disk after it was sputtered with Cu and the foil masks were removed.

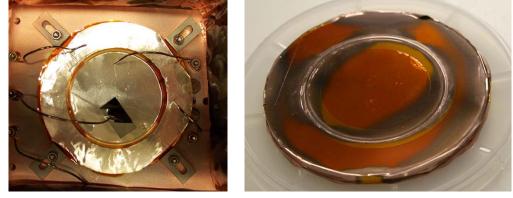


Figure 21. Setup for sputtering the Test Disk (left) and the sputtered disk after removing the Al foil masks (right)

While sputtering the Test Disk, a shutter in the sputtering chamber had been inadvertently left closed, so the first $0.5~\mu m$ of Cu was deposited very slowly, which had the effect of

significantly reducing the sputtering power. After sputtering the disk, the Kapton mask used during the sputtering process was left in place, and the disk was placed into the electroplating mask and electroplated with an additional several hundred μm of Cu. The disk was placed back on the diamond turning machine, and the Cu was machined with the diamond tool so that the Kapton could be removed without tearing the Cu. The electroplated Cu was then machined to a thickness of 43 μm . The Cu showed no signs of delaminating when attempting to pry up the edge with a razor blade. A picture of the machined Cu appears in Figure 22. Note that the band of Cu had a width of only 1 mm, because the design for the target changed. The edge of the Cu looked relatively clean, but there were a few places where the Al had been pitted adjacent to the Cu, which may have been a result of electroplating chemicals penetrating beneath the Kapton mask.



Figure 22. Machined Cu on the Test Disk with clean regions (left) and pitted regions (right)

SPUT1 - Masked with Electroplating Paint (10/05/04)

After successfully depositing a band of Cu on the Test Disk, an attempt was made to repeat the process on a precise 100 mm Al disk with parallel surfaces. This disk, labeled "SPUT1" was diamond turned using mineral spirits and cleaned with pentane, acetone, and ethanol. It was then masked with Bob Dively Liquid Masking Film and Kapton in the manner illustrated in Figure 20. The Kapton was scored with the diamond tool, and after peeling the 1 mm wide ring of Kapton tape and masking film from the surface, the machinist noticed a small piece of debris on the surface. In an attempt to clean off this debris, it was rinsed with ethanol. The alcohol appeared to have dissolved some of the adhesive on the Kapton tape and left a white residue on the 1 mm wide band of exposed Al. This residue was wiped with a cotton swab in an attempt to remove it.

The disk was then placed in the sputtering chamber in B231, ion milled for several minutes, and sputtered with 20 nm of Ti followed by several µm of Cu. Upon examining the disk with a microscope, it appeared that the Kapton tape had lifted slightly around the inner and outer perimeters of the 1 mm groove, as shown in Figure 23. This peeling of the Kapton may have been related to the fact that ethanol had dissolved some of the adhesive on the Kapton tape. It is possible that ethanol removed so much of the adhesive from the Kapton that there was not enough adhesive left to keep the Kapton down on the surface of the Al. There was concern that electroplating solution could easily have penetrated beneath the Kapton and damaged the Al surface of the disk, so the disk was not electroplated in this condition. Instead, the Kapton mask was removed, and the disk was coated with commercial electroplating masking paint. Special

care was taken to ensure that the edges of the 1 mm wide band of sputtered Cu were well covered.



Figure 23. Sputtered disk SPUT1 and a microscope image of the Kapton peeling up

Once the electroplating paint had dried, the disk was mounted back on the diamond turning machined. The paint was then scored with the diamond tool on either side of the 1 mm band of sputtered Cu, and this 1 mm wide strip of paint was peeled up to reveal the sputtered Cu, as shown in Figure 24.

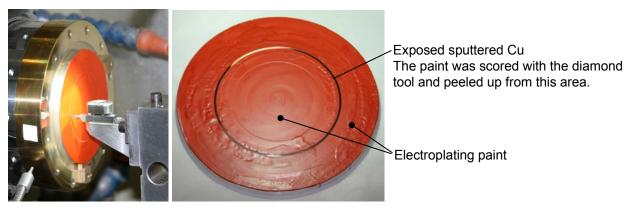


Figure 24. Scoring of the electroplating paint with the diamond tool (left) and the masked disk SPUT1 after peeling off the 1 mm wide section of scored paint (left)

The disk was then placed in the electroplating mask, and several hundred μm of Cu were deposited using a current density of 30 Amp/ft². The electroplating paint was then peeled off of the disk. The surface of the Al had been severely damaged during the electroplating process, as shown in the photograph and microscope image in Figure 25. The pitting of the Al was probably caused by electroplating chemical penetrating through the electroplating paint. Despite the fact that this electroplating masking paint is a commercial product that is commonly used to mask parts while they are being electroplated, it failed to adequately protect the Al disk.



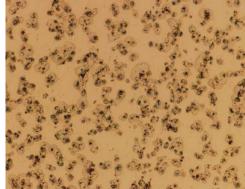
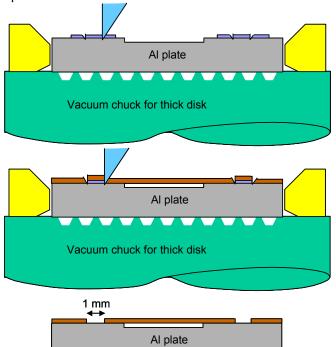


Figure 25. Damaged Al surface of SPUT1 caused by the electroplating process

SPUT3 - Masked with Kapton and Electroplating Paint (10/18/04)

Before processing another 100 mm Al disk, a simple test was performed to determine if electroplating chemicals can penetrate the commercial electroplating masking paint. The tests indicated that more than one coat of the electroplating paint is required to adequately protect the Al. Work continued with this electroplating paint, because it could be machined more nicely than Kapton tape, and at the time, it seemed like an attractive option that was expected to produce clean edges on the deposited Cu.

Another 100 mm Al disk, labeled "SPUT3" was diamond turned using mineral spirits and cleaned with pentane, acetone, and ethanol. The disk was prepared using the process illustrated in Figure 26. The disk was masked with Bob Dively Liquid Masking Film and Kapton. The Kapton was scored with the diamond tool, and the 1 mm wide ring of Kapton tape and masking film was peeled up. The disk was ion milled and sputtered in B231 with 20 nm of Ti followed by 5 µm of Cu.



- •Apply a layer of Bob Dively Liquid Masking Film over a portion of the disk.
- •Allow the masking film to dry for a few hours, and then score it with a diamond tool. The tool may cut up to 2 μm into the Al. •Peel up the outer and inner sections of paint, leaving only a 1 mm wide band of paint.
- •Apply Kapton tape over the entire surface of the disk
- •Score through the tape, being careful to retrace the previous score marks.

•Peel up the 1 mm wide ring of Kapton and masking film to expose a 1 mm wide band of Al

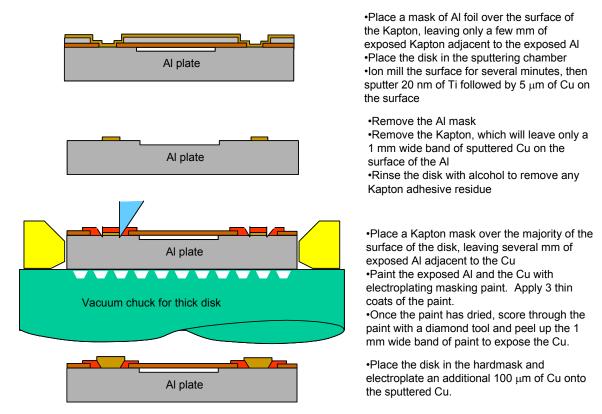


Figure 26. Processing of disk SPUT3

After sputtering the disk, the Kapton was peeled from the disk, and precut pieces of fresh Kapton were placed on the disk, leaving several mm of Al exposed adjacent to the sputtered Cu. The exposed Al and Cu were then covered with three coats of the commercial electroplating masking paint, as shown in Figure 27. Once the paint had dried, the disk was mounted back on the diamond turning machined. The paint was then scored with the diamond tool on either side of the 1 mm band of sputtered Cu, and this 1 mm wide strip of paint was peeled up to reveal the sputtered Cu.

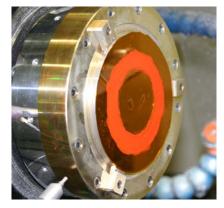




Figure 27. Disk SPUT3 after coating the Al and Cu with electroplating paint (left) and after scoring and peeling up the 1 mm wide section of paint that had covered the sputtered Cu (right)

The disk was then placed in the electroplating mask, and several hundred µm of Cu were deposited using a current density of 30 Amp/ft². The electroplating paint and Kapton were then peeled off of the disk. Once again, the surface of the Al was damaged during the electroplating process, as shown in Figure 28. The Al was pitted over portions of the area that had been covered by the electroplating paint, but it was clean in the area that had been covered by Kapton. The electroplating chemicals appeared to have leaked through or beneath the electroplating paint. At this point, it was obvious that despite the fact that the commercial electroplating paint was designed specifically for the purpose for which it was being used, it did not function well enough to meet the requirements for these targets.



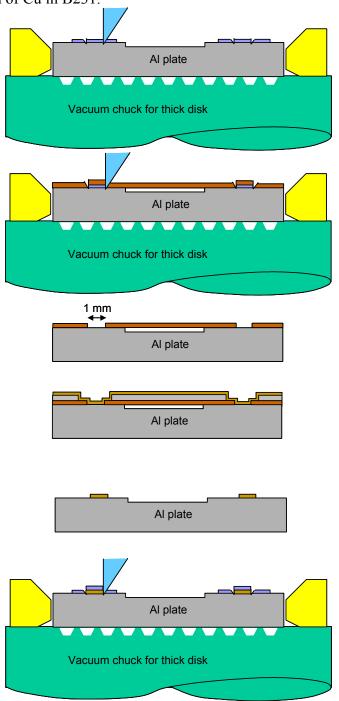
Figure 28. Damaged Al surface of SPUT3 caused by the electroplating process

Note that initial tests with the electroplating paint revealed that three coats of the paint adequately protected an Al substrate. However, when performing these tests, the Al was simply coated with the paint and then placed in the electroplating solution. The paint on this Al substrate was never machined with a diamond tool. In contrast, the paint on disk SPUT3 was machined with a diamond tool. It is possible that the process of machining the paint with the diamond cutting tool created a path for the electroplating chemicals to penetrate beneath the paint.

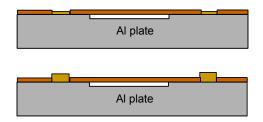
SPUT10 - Masked with Fresh Kapton Prior to Electroplating (11/03/04)

Because the electroplating paint did not function as expected, it was eliminated from consideration. Continued attempts to produce targets used Kapton tape to mask the 100 mm Al disk. Another 100 mm Al disk, labeled "SPUT10" was diamond turned using mineral spirits and

cleaned with pentane, acetone, and ethanol. The disk was prepared using the process illustrated in Figure 29. The disk was masked with Bob Dively Liquid Masking Film and Kapton. The Kapton was scored with the diamond tool, and the 1 mm wide ring of Kapton tape and masking film was peeled up. The disk was ion milled and sputtered with 20 nm of Ti followed by several µm of Cu in B231.



- •Apply a layer of Bob Dively Liquid Masking Film over a portion of the disk.
- •Allow the masking film to dry for a few hours, and then score it with a diamond tool. The tool may cut up to 2 μ m into the Al.
- •Peel up the outer and inner sections of paint, leaving only a 1 mm wide band of paint.
- •Apply Kapton tape over the entire surface of the disk
- •Score through the tape, being careful to retrace the previous score marks.
- •Peel up the 1 mm wide ring of Kapton and masking film to expose a 1 mm wide band of Al
- •Place a mask of Al foil over the surface of the Kapton, leaving only a few mm of exposed Kapton adjacent to the exposed Al
- •Place the disk in the sputtering chamber •lon mill the surface for several minutes, then sputter 20 nm of Ti followed by 5 μm of Cu on
- the surface
- •Remove the Al mask
- •Remove the Kapton, which will leave only a 1 mm wide band of sputtered Cu on the surface of the Al
- •Rinse the disk with alcohol to remove any Kapton adhesive residue
- •Repeat the procedure of applying Bob Dively Liquid masking film, scoring the dried film, and peeling up the inner and outer portions to leave a 1mm wide strip of masking film that covers the sputtered Cu.



- •Place a sheet of Kapton over the entire disk
- Score the Kapton along the score marks
- •Peel up the 1 mm wide strip of Kapton and masking film to expose the 1 mm band of sputtered Cu
- •Place the disk in the hardmask and electroplate an additional 100 μm of Cu onto the sputtered Cu.

Figure 29. Processing of disk SPUT10

Because disk SPUT1 had experienced problems with the Kapton peeling up from the surface of the Al after sputtering, disk SPUT10 was processed slightly differently. Future discoveries would later indicate that the problem with SPUT1 was caused by the fact that ethanol had dissolved some of the adhesive from the Kapton tape, and the problem was not caused exclusively by the nature of the Kapton itself. However, at the time that disk SPUT10 was processed, this information had not been discovered yet.

After sputtering Cu onto the disk, the Kapton was removed. Bob Dively Liquid Masking Film was then applied over the sputtered Cu. Once it had dried, the Bob Dively Liquid Masking Film was scored with the diamond tool, and a 1 mm wide strip was peeled up to reveal the sputtered Cu. A fresh layer of Kapton tape was then placed over the entire surface of the disk. The Kapton was then scored, and a 1 mm wide strip was peeled up to reveal the sputtered Cu. The disk was then placed in the electroplating mask, and several hundred μ m of Cu were deposited using a current density of 30 Amp/ft².

The Kapton was peeled off of the disk to reveal that once again, the Al was damaged adjacent to the electroplated Cu, as shown in Figure 30. It appeared that once again, chemicals from the electroplating process reached the Al and caused pitting of the surface. In this case, the chemicals probably penetrated beneath the edge of the Kapton tape.

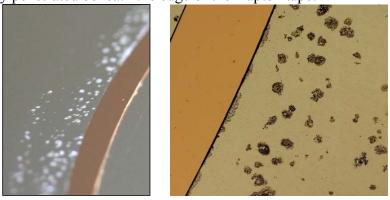


Figure 30. Pitting of Al on disk SPUT10 after electroplating

SPUT11 - Same Kapton Mask Used for Both Sputtering and Electroplating (11/19/04)

The only 100 mm Al disk that did not experience severe pitting during electroplating was the Test Disk. For this disk, the same Kapton mask was used for both sputtering and electroplating, and it did not peel up significantly during sputtering. The masking of the disk for the electroplating process had the configuration illustrated in Figure 31.

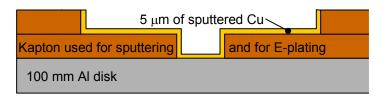


Figure 31. Masking of Test Disk during electroplating.

One possible explanation for why the Kapton did not peel up during the sputtering process was that the initial 0.5 µm of Cu was sputtered very slowly, so it did not heat the Kapton and cause it to lift from the surface. This bit of Cu sealed the Kapton to the Al, and subsequent Cu built up on the sidewall of the Kapton, which protected the Al disk from the electroplating chemicals. It may also have been relevant that the area onto which Cu was electroplated had a width of 11 mm on the Test Disk, but it was only 1 mm for SPUT1, SPUT3, and SPUT10.

The procedure used to prepare the Test Disk for sputtering was repeated for another disk, which was labeled SPUT11. This disk was machined using mineral spirits as the cutting fluid. It was then rinsed with pentane, acetone, and ethanol. The cleaning process was repeated three times. The disk was then coated with Bob Dively Liquid Masking Film, which was scored and peeled to leave only a 1 mm wide band on the surface of the disk. The entire surface of the disk was masked with Kapton tape, which was then scored with the diamond tool. The 1 mm wide ring of Kapton was peeled up to expose a 1 mm wide band of Al, as illustrated in Figure 20.

Prior to sputtering this disk, the edges of the Kapton were examined very carefully. Some images of the edge of the Kapton appear in Figure 32. Over the majority of the part, the Kapton appeared to be fairly well adhered to the Al, but there were some areas where the Kapton did not appear to be well adhered. The inspection also revealed that the diamond tool did not always cut a nice smooth edge in the Kapton, and burrs or stips of material could be created in the Kapton by the cutting tool.

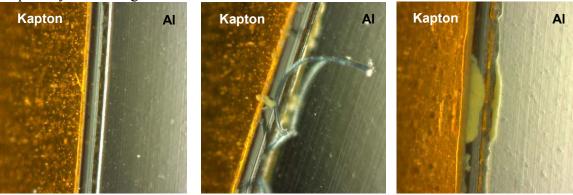
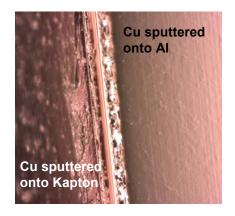


Figure 32. Disk SPUT11 masked with Kapton prior to sputtering

This part was placed in the sputtering chamber in B231, ion milled for several minutes, and then sputtered with 20 nm of Ti followed by 3 μ m of Cu. The deposition rate, which is a function of the sputtering power, was reduced to 50% of that used for the previous disks. Some images of the edge of the sputtered Kapton appear in Figure 33.



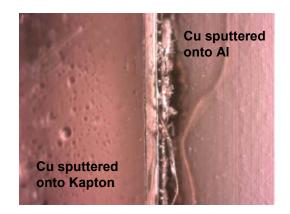


Figure 33. Disk SPUT11 after being sputtered with 3 μm of Cu

The disk was then placed in the electroplating mask, and approximately $100~\mu m$ of Cu was plated onto the sputtered Cu. The sputtering was done at $5~Amp/ft^2$ and required approximately 8~hours to deposit $100~\mu m$ of Cu. When the electroplating was complete, the Kapton was removed from the disk, and the Cu was diamond turned to a thickness of $45~\mu m$. Microscope images of the machined Cu and the Al appear in Figure 34. The majority of the Al surface had the original diamond turned finish. However, there were some areas in which the Al adjacent to the Cu had been pitted. Focusing the microscope in and out of the plane of the Al indicated that there may be some Al material near the pit that had been redeposited on top of the original Al surface. Therefore, it is not known how well this part would fit into a vacuum chuck with the Cu side against the face of the chuck.



Figure 34. Disk SPUT11 after diamond turning the electroplated Cu

Robin2 – Masked with Lithographic Photoresist that was Exposed with a Glass Photomask (11/23/04)

At the same time disk SPUT11 was being processed, an alternate means of masking 100 mm Al disks was being pursued. This alternate method used a lithographic photoresist to mask the part while sputtering the Cu. Lithographic masks are known to produce edges with sub-µm accuracy when sputtering or evaporating layers of materials with a thickness of several dozen nm. However, it was not known at the time of this effort how these masks would perform when attempting to make EOS targets via sputter-seeded electroplating of Cu.

Another 100 mm Al disk, labeled "Robin2" was diamond turned using mineral spirits. Two fiducial marks were then machined onto the surface of the disk. The disk was then rinsed with pentane, acetone, and ethanol. The cleaning process was repeated three times. 14 µm of

lithographic photoresist was spin-coated onto the surface of this disk in B153, and the resist was then soft-baked. The disk was then placed into a machine that aligned a glass photomask with the fiducial marks that had been machined on the disk. After exposing and developing the photoresist, it was hard-baked. This process removed a ring of the photoresist, so that a 1 mm wide ring of Al was exposed through the photoresist, as shown in Figure 35.

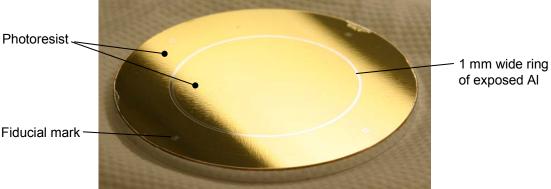


Figure 35. Disk Robin2 masked with photoresist

The exposed Al was measured with the Wyko NT8000 optical profiling system using PSI mode at $20\times$. Some features were apparent on the exposed Al, as shown in Figure 36. The lithographic process appeared to have changed the surface finish of the Al from S_t =40-50 nm to S_t =90-100 nm when measured in PSI mode at $20\times$.

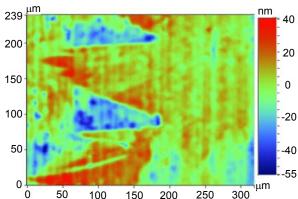


Figure 36. Wyko measurement of exposed Al on disk Robin2

The disk was placed into the sputtering chamber, where it was ion milled for several minutes and then sputtered with 3.8 μ m of Cu at 50% power. When the disk was taken out of the sputtering chamber, the photoresist had curled up and delaminated from the Al, as shown in Figure 37. The photoresist may not have maintained adhesion to the Al due to the heat from the sputtering process. It is also possible that the photoresist was not rigid enough to withstand the residual stress formed in 3.8 μ m of sputtered Cu. Photoresists such as this are normally used to sputter only a few dozen nm of material, and a coating of 3.8 μ m may have simply been too thick for the resist.

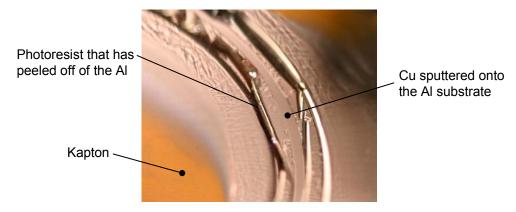
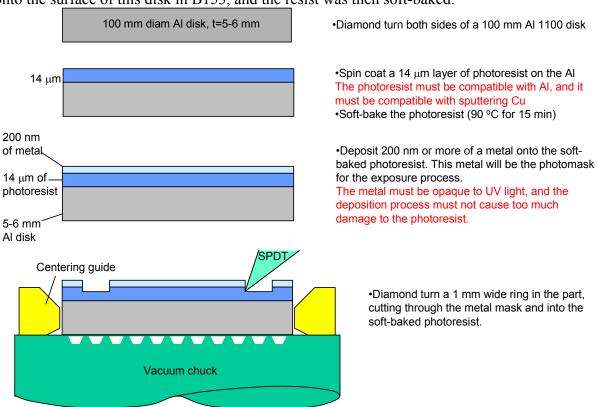


Figure 37. Delaminating photoresist on disk Robin2 after sputtering 3.8 µm of Cu onto the photoresist

Robin1 – Masked with Lithographic Photoresist that was Exposed with an E-Beam Evaporated Cr Photomask (12/10/04)

At the same time disk Robin2 was being processed, another 100 mm Al disk was being prepared using a slightly different type of lithographic masking process. This different masking process is illustrated in Figure 38. This other 100 mm Al disk, labeled "Robin1" was diamond turned using mineral spirits. The disk was then rinsed with pentane, acetone, and ethanol. The cleaning process was repeated three times. 14 μ m of lithographic photoresist was spin-coated onto the surface of this disk in B153, and the resist was then soft-baked.



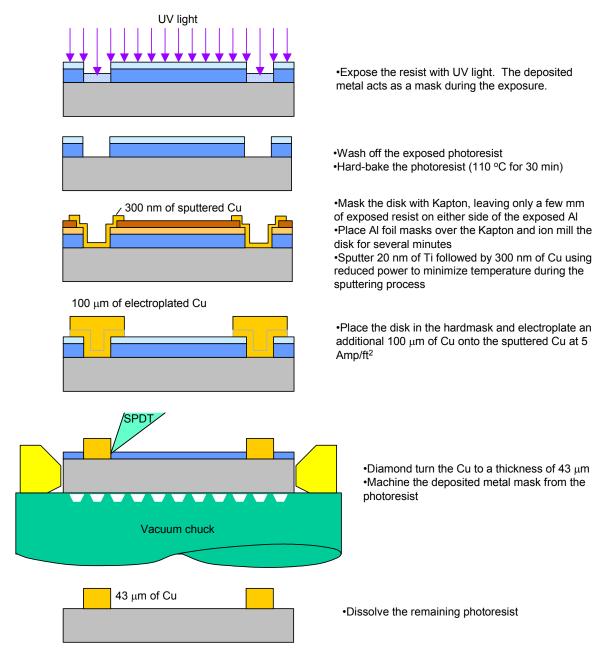


Figure 38. Processing of disk Robin1

200 nm of Cr was then evaporated onto the surface of the photoresist. After evaporating the Cr onto the resist, the surface began to crack and looked like a dried lake bed. The disk was placed back on the diamond turning machine, and a groove of width 1 mm was machined through the Cr and into the underlying resist. Microscope images of the 1 mm groove in the Cr appear in Figure 39. The cracked surface of the Cr is clearly visible in these images.

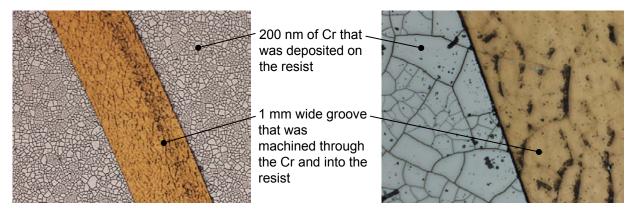


Figure 39. Surface of disk Robin2 after machining a 1 mm groove through the 200 nm of Cr on the resist

The disk was then placed under a UV light source to expose the resist. Note that the 200 nm of Cr acted as a photomask during the exposure, so only a 1 mm section of the photoresist was exposed to the UV light. The exposed resist was then washed off. The remaining photoresist was then hard-baked, and the disk was masked with Kapton. The disk was then ion milled for several minutes and sputtered with 20 nm of Ti followed by 300 nm of Cu in B231. In this case, the photoresist did not delaminate from the Al substrate. The disk was then placed in the electroplating mask, as shown in Figure 40.

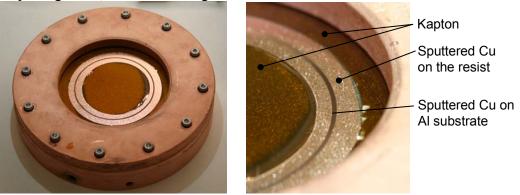


Figure 40. Disk Robin1 in the electroplating mask

The disk was then electroplated at 5 Amp/ft² for 8 hours. The disk was placed back on the diamond turning machine, and the excess electroplated Cu was machined off with the diamond tool. The disk was then rinsed with acetone to remove the remaining photoresist.

After washing away all of the photoresist with acetone, islands of Cu were evident on the Al. The Al was not pitted as it had been with SPUT1, SPUT3, and SPUT10, and it was not damaged near the edge of the mask as it had been with SPUT11. Instead, the Al surface was partially covered with small islands of Cu. The images in Figure 41 were taken after machining the Cu islands to a height of approximately 2 μ m from the surface of the Al. The 1 mm Cu band had a height of 50 μ m.

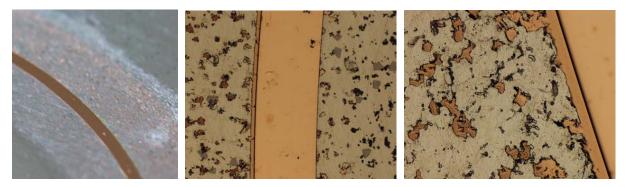


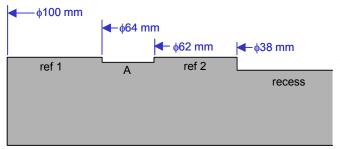
Figure 41. Robin1 after electroplating and removing the lithographic mask (center=25×, right=100×)

The islands of Cu on the Al could have resulted from the cracks in the Cr photomask and underlying photoresist. It is possible that during the sputtering process, Cu penetrated through the Cr and the photoresist (possibly through the cracks) and reached the Al surface. Additional Cu was then deposited during the electroplating processes to create the islands seen in Figure 41. Note that if the photoresist had not contained the cracks seen in Figure 39, this process might have been successful. However, at this point in the development process, resources were not available to make another attempt with a lithographic mask.

D1, D2, and D3 – Cu Deposited into a Machined Recess in the Al Disk (12/13/04)

Rather than attempting to avoid pitting of the 100 mm Al disk adjacent to the Cu during electroplating, a process was developed to remove the pitted material from the Al so that it would not be present on the finished targets. This change in direction in the development of the manufacturing process resulted from a relaxation of some of the specifications by the physicists. The original specifications for the targets required that the Al baseplate have no short-wavelength features with a peak-to-valley height larger than 0.1 μ m. This specification was relaxed to allow a step of up to 1 μ m in the Al baseplate at the edge of the Cu sample, provided the height of any step was measured to better than 0.2 μ m. This change in the specifications allowed a completely different means of depositing and machining the Cu sample to be pursued.

Three additional 100 mm Al disks were prepared. After diamond turning both sides of each disk and cleaning them with pentane, acetone, and ethanol, a specific profile was machined into the surface of each disk. This profile included a recess of width 1 mm and depth 25 μ m. This recess was carefully measured with an LVDT mounted to the B-axis of the diamond turning machine. After masking each disk with Kapton tape, several μ m of Cu was sputtered into the 1 mm \times 25 μ m recess, and additional Cu was then electroplated on the sputtered Cu, as described in Figure 42.



Surfaces ref1 and ref2 are reference surfaces. Surface A is machined 25 μm below the reference surfaces. The Cu will be sputtered onto this surface.

The recessed surface is machined 58 μm below the reference surfaces and will be used to make the Al characterization targets.

Depth of A = 25 μ m Depth of recess = 58 μ m

Using LVDT, measure actual •Depth of A below ref2 •Depth of recess below ref2

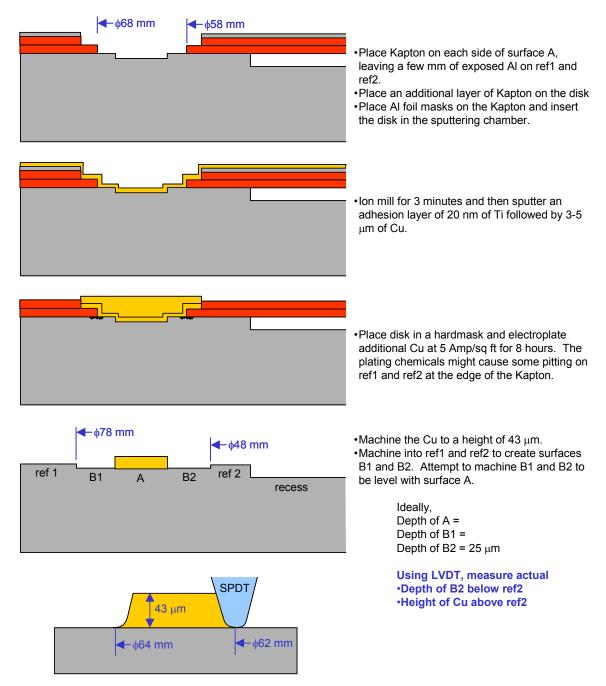


Figure 42. Processing of D1, D2, and D3 by sputter-seeded electroplating the Cu into a recess in the Al

When the electroplated Cu was diamond turned to its final form, the pitted Al adjacent to the Cu was machined at the same time. A special method was developed to diamond turn surfaces B1 and B2 coplanar with the interface of the Al and the Cu (surface A in Figure 42). This method involved making several intermittent measurements of the Al using the LVDT mounted to the B-axis of the diamond turning machine. By intermittently measuring the step height between surface B2 and the reference surface ref2, the amount of material remaining to be removed could be determined, as shown in Figure 43. The final machining pass had a depth of cut of 2 μ m. Surface A was measured to be flush to surface B2 to within approximately 20 nm, where the uncertainty of this measurement was approximately ± 50 nm.

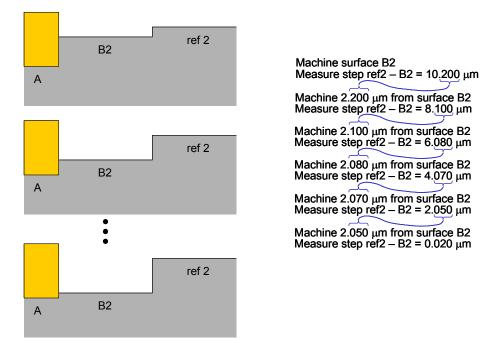


Figure 43. Method for machining the Al flush with the bottom surface of the Cu

Photographs of disk D2 at various stages in the process appear in Figure 44. Unfortunately, the diamond tool was mistakenly plunged into the Cu of disk D1, which ruined this disk. Therefore, processing continued using disk D2.

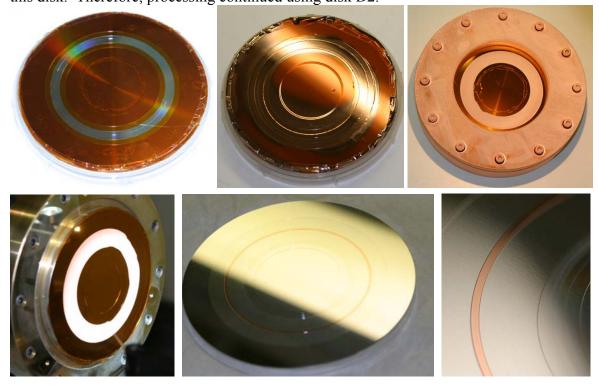


Figure 44. Photographs of disk D2 after being masked with Kapton (top left), sputtered with Cu (top center), placed in the hardmask for electroplating (top right), electroplated with additional Cu and mounted on the vacuum chuck (bottom left), and diamond turned (bottom center and bottom right)

Once the 1 mm band of Cu had been deposited onto the surface of the Al disk and diamond turned to a thickness of 43 µm, a vacuum chuck was fabricated to hold the 100 mm Al disk so it could be machined from a thickness of 6 mm to 62 µm. This vacuum chuck included a 1 mm × 43 µm recess for the band of Cu, and a 58 µm step, so the vacuum chuck would match the profile of the Al disk. To reduce the likelihood of the thin Al pulling away from the vacuum chuck during machining, the Al was reduced to a thickness of 62 µm only in the regions of the disk in which the targets would be made. In these regions, the vacuum chuck was diamond turned with vacuum grooves of width 100 µm and lands of width 50 µm. Elsewhere on the disk, the Al was left thicker, so it would have additional structural stability. In these regions, the grooves on the vacuum chuck had a larger width, but all of the lands on the chuck were 50 µm wide. The Al around the perimeter of the 100 mm disk was left at a thickness of 6 mm, so the disk could be handled easily. The profile that was machined into the Al disk is illustrated in Figure 45. The purple numbers in the figure indicate the thickness of the Al in the different regions, and the red numbers indicate the width of the grooves used in each region of the vacuum chuck. The vacuum grooves were connected together by eight radial grooves, which were machined by placing a diamond tool on its side and machining linear grooves from the inside toward the outer perimeter of the chuck. These radial grooves were intended to have a depth of 25 μm.

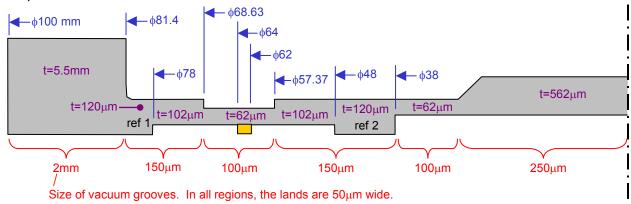


Figure 45. Schematic illustration of the profile machined into the 100 mm Al disk and the widths of the grooves on the vacuum chuck. The figure is not drawn to scale.

Several weeks were required to plan the fabrication of this chuck, write the machine tool programs, debug the programs, machine the vacuum grooves, cut radial channels to connect each of the vacuum grooves, and reface the chuck.

Upon placing one of the disks into this vacuum chuck, the axial runout of the disk measured by the capacitance probe was approximately 1.5 µm. However, after squirting the perimeter of the disk with alcohol, the axial runout dropped to 200 nm. Apparently, the alcohol sealed off the perimeter of the disk, so the vacuum could pull the entire disk flush against the vacuum chuck, as shown in Figure 46.

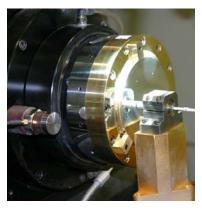


Figure 46. Cap probe measuring axial runout on disk D2 with Cu against the vacuum chuck

When machining the Al disk from a thickness of 6 mm to thin foil, a HSS tool was used to make the initial cuts. HSS was used because it can be ground to have a sharper cutting edge than a carbide tool. When spray mist mineral spirits are used as a cutting fluid, the Al does not build up on the tool, and the HSS has adequate tool life. After removing several mm of Al, the HSS tool was replaced with a diamond tool of nose radius $125~\mu m$, which machined the Al with a depth of cut of $1.5~\mu m$. Note that a smaller depth of cut caused the Al to burnish rather than producing clean chips.

When the Al had reached a thickness of 90 µm, a failure occurred. The workpiece began making an odd noise, so the machine tool program was halted. The tool had cut through the Al at the outer edge of the Cu. However, the cut did not extend a full 360° around the part. One damaged area extended over an included angle of approximately 60°, over which the Al baseplate appeared to have been cut cleanly by the diamond tool. Over another 40° to 50° length, a portion of the Cu appeared to have been ripped off by the diamond tool.

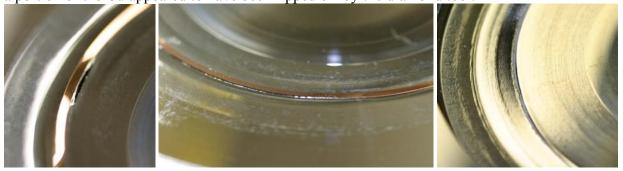


Figure 47. Disk D2 after being machined to a thickness of 90 µm. Small rip on inside of Cu (left), large clean cut through outside of Cu (center), large rough cut 1 mm inside the Cu on the machined face of the Al (right)

There were also some damaged areas on the inner edge of the Cu. One cut extended over a fairly short arclength, as shown in the left side of Figure 47. The Al was also cut in several places several mm inside the Cu. None of the damaged areas extended a full 360° around the part.

The cause of this failure was traced to the radial grooves on the vacuum chuck in the location of the outer edge of the 1 mm \times 43 μ m Cu band (the same location where disk D2 first failed while being machined to thickness). It was discovered that the radial grooves in the outermost land that supported the Cu were incomplete and did not extend fully through the land. Therefore, there was no continuous path for the vacuum to reach outside this land. The clearance between the bottom surface of the Cu and upper surface of the lands in the 43 μ m recess in the

vacuum chuck was nominally only 50 nm. Therefore, because the radial grooves were incomplete and did not properly create a vacuum path through this particular land, any vacuum feed would be restricted to this 50 nm clearance. Note that because of uncertainties in the process and in the measurements, the Cu could possibly have actually bottomed out on these lands, in which case there would actually have been no clearance and no path for the vacuum. The thin Al may have pulled away from the vacuum chuck, where it was severed by the diamond tool.

A diamond tool was placed on its side to machine radial grooves in the vacuum chuck that connect each of the vacuum channels. The tool machined from the center to the perimeter of the chuck.

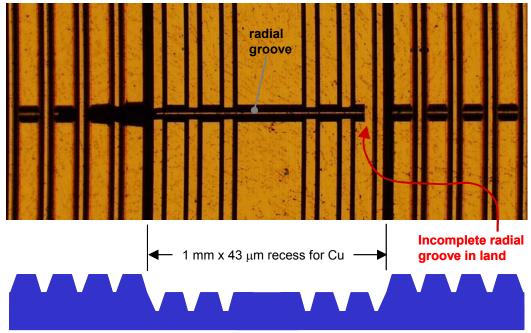


Figure 48. Error on the vacuum chuck used to machine disk D2

This problem with the vacuum chuck was corrected before machining disk D3. When disk D3 was placed against this vacuum chuck, its axial runout was 150 nm. This disk was successfully machined to a thickness of 63 μ m, as shown in Figure 49. Note that the thickness of 63 μ m was within the specification of 62±2 μ m.

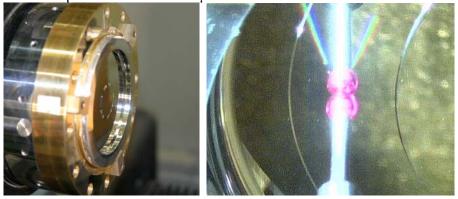


Figure 49. Disk D3 after being machined to a thickness of 63 µm (left) and measured with the LVDT (right)

The cut schedule used when machining the Al to a thin foil is listed in Figure 50.

	Tool Nose Radius		Starting Al	Ending Al thickness
Tool	[µm]	Depth of Cut [μm]	thickness [µm]	[µm]
HSS	250	25	6,000	2,000
Diamond	125	20	2,000	500
Diamond	25	10	500	200
Diamond	25	1.5	200	63

Figure 50. Cut schedule used when machining the Al to a thin foil

BONDING THE ICH AND CH PADS TO THE 100 mm AI DISK

Development of a bonding process

Prior to machining the 100 mm Al disks to a thickness of 62 μ m, testing was performed to develop a method of bonding the ICH pads for the pre-heat shields and the CH pads for the ablators with an adhesive thickness of less then 3 μ m. Several ICH and CH pads with a thickness of 500 μ m had been fabricated with the forms shown in Figure 51. Originally, ICH and CH pads with the form shown on the left side of Figure 51 were fabricated. Both surfaces of these pads were diamond turned. However, additional pads had to be fabricated, and the new pads had the form shown on the right side of Figure 51. These pads were fabricated using a slightly different process that was significantly faster and less expensive.

To fabricate the pads shown on the right side of Figure 51, a single point diamond tool, a diamond parting tool, and a HSS parting tool were setup on a two-axis diamond turning machine. The pads were machined from a rod of ICH or CH mounted in the spindle. The single point diamond tool machined the face of each pad, including the 3° chamfer. The diamond parting tool then parted most of the way through the rod, machining the outer portion of the back face of the pad in the process. Note that the diamond parting tool created a diamond turned surface that was parallel to the front face of the pad to better than $0.1~\mu m$. However, the diamond parting tool was too short to reach the center of the pad, so the remainder of the parting operation was performed with a HSS parting tool, which resulted in a poor surface finish on the center section of the back face.

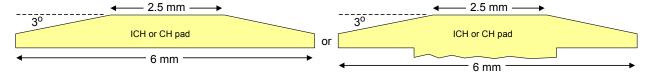


Figure 51. Two different types of ICH and CH pads used for the EOS targets

Several of the machined pads were measured on the Wyko using PSI mode at $20\times$. The Rexolite pads had a surface finish of approximately Rt=70-80 nm, Ra=5-6 nm, Rq=7 nm. The ICH pads had a surface finish of approximately Rt=70-80 nm, Ra=4 nm, Rq=5 nm. Therefore, the surfaces of the pads were well within the specification of Rt < 600 nm.

The thickness and thickness uniformity of the ICH and CH pads were measured by placing them on a vacuum chuck on the diamond turning machine and measuring them with an LVDT. Figure 52 depicts the measurement of one of the CH pads illustrated on the left side of Figure 51.



Figure 52. Measurement of thickness and thickness uniformity of a CH pad

The accuracy of this method of measuring the pads was determined by measuring a gauge block with a thickness that was certified to be within 50 nm of 3 mm. This block was wrung onto the flat section of the vacuum chuck and was measured to be within 10 nm of 3 mm. The gauge block was also placed in the center of the vacuum chuck (so that it was being held by vacuum) and measured again in six places. All six measurements were within 50 nm of 3 mm. Therefore, when measuring the thickness of a rigid part, this method is probably accurate to approximately ± 50 -100 nm.

The pads depicted on the right side of Figure 51 had to be measured using a special vacuum chuck that provided clearance for the rough material near the center of the back face. It was later discovered that these ICH and CH pads flexed from the vacuum pressure while they were being measured with the LVDT. Each of the pads may have been measured with an error of 0.6 to 1.0 μ m. While measuring each of the ICH and CH pads, a 4.375 mm diameter section of the pad was unsupported by the vacuum chuck, as shown in Figure 53.

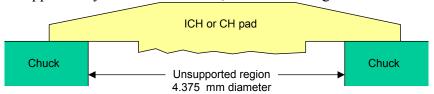


Figure 53. Support of an ICH or CH pad with a vacuum chuck while its thickness is being measured

An finite element analysis indicates that the vacuum pressure of 50,796 Pa (15 in Hg) would have caused the center section of the pad to deflect 0.6 to $1 \mu m$ (assuming the modulus of the plastic is 1.8 GPA to 3 GPA, respectively), as shown in Figure 54. Therefore, the recorded values of the pad thickness may have been 0.6 to $1 \mu m$ less than the actual pad thickness.

After bonding the pads to the 100 mm diameter Al disk, this deflection will not occur. Therefore, the if the raw measured values of the pad thickness are used to calculate the adhesive layer thickness, then the adhesive layer thickness will appear to be 0.6 to 1 μ m larger than it actually is.

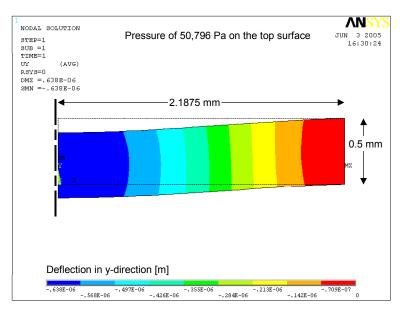


Figure 54. Finite element analysis of the deflection of an ICH or CH pad while its thickness is being measured

To avoid this problem, the back surface of the pad should be machined using a diamond parting tool that is long enough to reach the center of the pad. If the entire back surface of the pad is diamond turned, then the entire back surface of the pad could be supported by a vacuum chuck while its thickness is being measured. Alternatively, a low vacuum pressure could be used when measuring the thickness of the ICH and CH pads with the LVDT.

UV curing adhesives

To practice bonding the ICH and CH pads onto the 63 µm thick Al, several pads were bonded to a diamond turned aluminum flat using various adhesives. The first adhesives tested were UV adhesives, which cured upon exposure to ultraviolet (UV) light. These adhesives are available in a range of viscosities, and they do not cure until they are exposed to UV light for several seconds. Tests were performed with LocTite 3104, 3494, and 3525.

The bonding tests were performed directly on the diamond turning machine. A 100 mm diameter Al workpiece was diamond turned and left on the machine tool, and the air bearing assembly station shown in Figure 55 was mounted on the B-axis of the machine tool. This assembly station incorporates an air bearing in series with a spring and a force transducer and allows the operator to control the assembly force to within approximately 0.01 N.

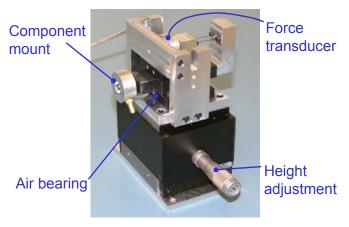


Figure 55. Air bearing assembly station used to bond the ICH and CH pads

One of the pads was bonded onto the tip of a polystyrene assembly cone attached to one of the aluminum holders shown in Figure 3. The pad was bonded to the cone using a small droplet of LocTite 3103 UV curing adhesive, as shown in the top left picture of Figure 56. The assembly of the pad onto the cone was performed with a benchtop assembly station. The aluminum holder containing the cone and the pad was then placed on the assembly station on the diamond turning machine. Using a small brush (for the LocTite 3104) or a single hair (for the LocTite 3494 and 3525), a drop of UV curing adhesive was placed onto the pad using a hair, as shown in the top center picture of Figure 56. The pad was then pressed against the Al flat with a force of 5 grams-force. The C-axis of the DTM then moved the pad through an arclength of 0.75 mm in either direction and then back to the original position, as shown in the top right picture of Figure 56. Note that the C-axis of the Moore 350 diamond turning machine does not obey feedrate commands. In order to control the rate of motion of the pad with respect to the Al, a 5 um move in the x-direction accompanied the rotation of the C-axis. The pad was then pressed further into the Al until the force reached 10 grams-force, and the C-axis moves were repeated. The adhesive was then cured using the handheld Norland UV light, as shown in the bottom right picture of Figure 56. The UV light was held approximately 13 mm from the part for 2 minutes.

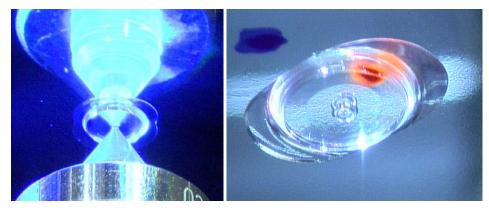


Figure 56. Bonding of an ICH pad to an assembly cone, application of UV curing adhesive, application of the pad to an Al flat, and curing of the adhesive with a UV light source

The assembly cone was freed from the pad by rotating the C-axis. An image of the bonded pad appears in the bottom right picture of Figure 56. The LVDT mounted to the B-axis platform was then used to measure the distance from the surface of the bonded pad to the surface of the Al flat in four spots. The thicknesses of the adhesives in the four measured locations are given below. The pads that were used to perform these initial bonding tests were not subject to the measurement uncertainty illustrated in Figure 53, so these numbers reflect the actual thickness of the adhesive in the measured location with an uncertainty of approximately 100 nm.

LocTite 3525	(10,000 cP)	Force=15g	0.13	2.10	1.20	1.86 μm
LocTite 3104	(100 cP)	Force=10g	0.14	0.69	0.51	0.23 μm
LocTite 3494	(6,000 cP)	Force=10g	2.27	1.56	1.50	1.75 μm

These pads were then diamond turned to a thickness of approximately $100~\mu m$ using a diamond tool with a nose radius of $125~\mu m$. This operation revealed that the surfaces of the UV adhesives remained tacky after being cured by the UV light. The tackiness probably resulted from oxygen in the air inhibiting the curing process near the surface of the adhesive. Therefore, chips generated in the diamond turning process were also tacky and adhered to the freshly diamond turned plastic and the adjacent adhesive, as shown in Figure 57. It is important to note that the lighting conditions can significantly affect the visibility of debris on the surface of the pad. The pad might appear clean with certain lighting conditions, but debris can become apparent with a small change in the angle of incidence of the lamp, as shown in the picture on the right side of Figure 57.



Figure 57. Chips sticking to the diamond turned UV curing adhesive and the ICH pad

This debris could not be removed using a dry cotton swab. In an effort to free the debris, a swab containing alcohol was brushed over the surface of the pad. The alcohol appeared to partially dissolve the UV adhesive, which attracted additional debris onto the surface of pad. Therefore, the alcohol actually made the situation worse. Because of the difficulty in removing this debris from the surface, UV curing adhesives were abandoned in favor of other adhesives that were known to have more favorable curing properties.

Cyanoacrylate

A pad was bonded to the Al flat using LocTite 495 cyanoacrylate. This adhesive is commonly used by the Target Fabrication Group and was known to be fairly hard upon curing, and cured cyanoacrylate had often been machined with a diamond tool. Unfortunately, the geometry of this assembly operation prevented the cyanoacrylate from curing as expected. After applying a droplet of cyanoacrylate to an ICH pad and assembling the pad onto the Al flat, the pad was moved in an X pattern. After about 60 seconds, the cyanoacrylate had not cured, and the pad pulled off of the Al when the assembly cone was retracted. Another pad was then assembled using LocTite 495. After several minutes, the pad was firmly secured to the Al flat, but the cyanoacrylate in the chamfered section of the pad remained liquid and did not cure. Apparently, the cyanoacrylate did not cure completely, because it was not exposed to the atmosphere, where the humidity could cause it to cure. The cyanoacrylate in contact with the surfaces of the pads probably cured quickly because of the moisture resident on the surfaces. However, the thickness of the adhesive layer was apparently too large to achieve a completely cured layer.

Two-part epoxies

Two-part epoxies, such as Hardman Green and StyCast, do not experience problems with oxygen inhibition, and they do not require moisture to cure. Furthermore, they had been used in the past by the Target Fabrication Group and were known to have reasonably good diamond turning properties. Therefore, an additional test was performed to bond an ICH pad to the Al flat using Hardman Green epoxy. The epoxy was mixed in a small plastic vial with a plastic rod. It was then centrifuged for a few minutes, which removed all of the visible bubbles created during the mixing process. A droplet of this epoxy was then placed on an ICH pad, and the pad was assembled with a force of 10 grams-force and moved back and forth a distance of 0.75 mm across the surface of the Al, as shown in Figure 58.

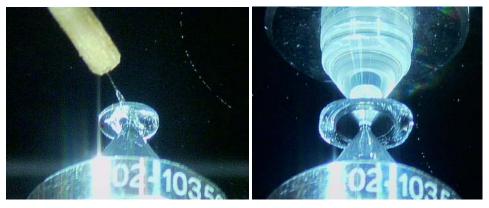


Figure 58. Application of Hardman Green epoxy to an ICH pad using a hair (left) and bonding of the pad to an Al flat (right)

The LVDT then measured the thickness of the adhesive in four locations. The measured values were 1.27, 2.18, 1.04, and 2.46 μm . Therefore, the epoxy successfully created a bond layer with a thickness of less than 3 μm .

The bonded ICH pad was then diamond turned to a thickness of approximately $100 \, \mu m$ using a diamond tool with a nose radius of $125 \, \mu m$. Images of the diamond turned pad appear in Figure 59. There were only a few pieces of debris visible on the surface, which could be removed using a cotton swab.



Figure 59. Machined ICH pad bonded to Al with Hardman Green epoxy (left). Microscope image of a typical diamond turned pad taken with Interference Contrast mode (right).

A CH pad was then bonded on top of this ICH pad using StyCast epoxy. StyCast has a relatively low viscosity and is often used to obtain thin bondlines. StyCast comes in pint-sized containers, and the two parts must be measured using a balance to obtain the proper mixing ratio. A 2 mL batch of this epoxy was mixed in a plastic vial and then centrifuged for several minutes. A few droplets of the epoxy were then applied to a CH pad using a hair. This epoxy has a low viscosity, so one must be careful not to apply too much of it to the pad. The pad was then assembled on top of the ICH pad on the Al flat. StyCast epoxy has a pot life of several hours, and it must cure for 7 to 8 hours before the assembly cone can be removed. Figure 60 illustrates the application of the StyCast to the CH pad, the bonding of the CH pad to ICH pad on the Al flat, and the bonded CH pad after removing the assembly cone.

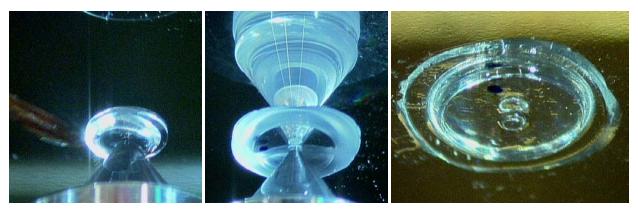


Figure 60. Bonding a CH pad on top of an ICH pad using Stycast epoxy

The LVDT then measured the thickness of the adhesive in four locations. The measured values were 1.07, 0.31, 0.66, and 0.69 μm . Therefore, the adhesive successfully created an adhesive layer well within the limit of 3 μm .

The CH pad was then diamond turned to a thickness of approximately $32.5~\mu m$. The StyCast epoxy did not machine as cleanly as the Hardman Green epoxy. The chips formed by the diamond tool had a tendancy to stick to the CH, but they could be removed using a cotton swab with some effort.

This study of bonding processes for the ICH and CH pads concluded that Hardman Green epoxy should be used to bond the ICH pad to the Al. A sub-µm bond can be created with StyCast epoxy, so StyCast could be used to bond the CH pads to the ICH pads. Note that Hardman Green could also be used to bond CH pads to the ICH pads. Hardman Green cures more quickly than the StyCast, but it has a larger viscosity and forms slightly thicker adhesive layers. A summary of the procedure for bonding the ICH and CH pads appears in Appendix C.

Bonding ICH and CH pads to disk D3

After machining disk D3 to a thickness of 63 μ m, it was kept on the vacuum chuck on the diamond turning machine, and six ICH pads were bonded to it. The center of each pad was placed at a radius of 31 mm, so it would lie directly over the inside edge of the 1 mm \times 43 μ m Cu band on the opposite face of the disk. Each ICH pad was bonded with Hardman Green epoxy using the process described in Appendix C. After bonding each of the pads, the step height from the surface of the pad to the Al baseplate was measured with the LVDT mounted to the diamond turning machine, as shown in Figure 61.



Figure 61. Measurement of adhesive layer thickness for an ICH pad

The LVDT measurements indicated that several of the adhesive bonds met the specification, but others were thicker than the limit of 3 µm. Many of the ICH pads contained debris that had to be removed prior to bonding. This debris was very difficult to see and was often only visible in certain lighting conditions. The thicker adhesive bonds probably resulted from debris that prevented the pad from seating correctly onto the Al baseplate. **Prior to bonding the pads in place, it is very important that they be adequately cleaned.** All debris should be removed from the surface.

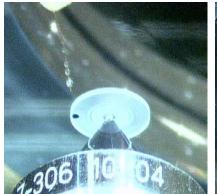
A summary of the adhesive bonds for the ICH pads appears in Figure 62. The values listed in the second, third, fourth, and fifth columns of this table were obtained by measuring the step height from the surface of the bonded pad to the Al baseplate and subtracting the measured thickness of the ICH pad. This step height was measured in four different spots around the perimeter of the 6 mm diameter ICH pad. The pad thicknesses used to make the calculations were the raw measurements made by the LVDT; the deflection of the pad while its thickness was being measured was not considered. The sixth column of the table interpolates these four calculated values to determine the thickness of the adhesive layer in the area of interest on the target. Once again, the deflection of the ICH pad while its thickness was being measured is not considered in this calculation. The final column of the table lists the likely actual thickness of the adhesive layer in the area of interest after the issue related to the measurement of the thickness of the ICH pad described in Figure 53 and Figure 54 has been considered.

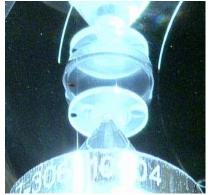
	Do	o not ac	count fo	Account for the deflection of the pad while its thickness was measured		
	Step1- Step2- Step3- Step4-					
	meas	meas	meas	meas	Calculated adhesive thickness	Likely actual adhesive
Pad	pad t	pad t	pad t	pad t	in area of interest	thickness in area of interest
ICH1	4.4	2.9	6.6	0.3	3.5	2.5 to 2.9
ICH2	4.1	1.4	3.6	1.9	2.8	1.8 to 2.2
ICH3	2.1	4.6	2.8	3.9	3.4	2.4 to 2.8
ICH4	3.3	2.9	2.9	3.6	3.2	2.2 to 2.6
ICH5	3.5	4.0	3.6	4.2	3.8	2.8 to 3.2
ICH6	3.4	2.0	3.1	2.7	2.8	1.8 to 2.2

Figure 62. Summary of adhesive bonds for the ICH pads. All numbers have units of µm.

After bonding the last ICH pad in place, the six pads were diamond turned to the proper thickness using a diamond tool with a nose radius of 25 μ m. A depth of cut of 1 μ m, a spindle speed of 1600 rpm, and a feedrate of 5 mm/min were used for the finishing passes. The LVDT verified that the thickness of each of the preheat shields, which consists of the ICH material and the Hardman Green epoxy, was 103.6 μ m.

A CH pad was then bonded directly over each ICH pad with StyCast epoxy using the procedure described in Appendix C. Figure 63 depicts the application of the StyCast to the CH pad, the centering of the CH pad over the ICH pad, and the application of the CH to the ICH. Each CH pad was allowed to cure overnight before removing the assembly cone.





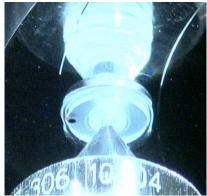


Figure 63. Bonding and assembly of a CH pad on an ICH pad on disk D3

The step height from the surface of each CH pad to the Al baseplate was measured with the LVDT mounted to the diamond turning machine. Once again, several of the adhesive bonds met the specification, but others were thicker than the limit of 3 μ m. The cleanliness of the CH pads and the surface of the ICH were critical to achieving acceptable adhesive layers, and debris on these surface probably prevented some of the pads from being bonded correctly. A summary of the adhesive bonds for the CH pads appears in Figure 64.

Figure 64 has the same form as Figure 62. The values listed in the second, third, fourth, and fifth columns of this table were obtained by measuring the step height from the surface of the bonded CH pad to the Al baseplate and subtracting the 103.6 µm thickness of the preheat shield and the measured thickness of the CH pad. The pad thicknesses used to make the calculations do not consider the deflection of the CH pad while its thickness was being measured. The sixth column of the table interpolates these four calculated values to determine the thickness of the adhesive layer in the area of interest on the target. Once again, the deflection of the CH pad while its thickness was being measured is not considered in this calculation. The final column of the table lists the likely actual thickness of the adhesive layer in the area of interest after the issue related to the measurement of the thickness of the CH pad described in Figure 54 has been considered.

	Do not	t for the	Account for the pad wh	ile it	_			
	Step1- Step2- Step3- Step4-							
	103.6 -	103.6 -	103.6 -	103.6 -				
	meas	meas	meas	meas	Calculated adhesive thickness	Likely ac	tual	adhesive
Pad	pad t	pad t	pad t	pad t	in area of interest	thickness in	are	a of interest
REX1	2.0	2.8	3.3	1.3	2.4	1.4	to	1.8
REX2	2.7	3.1	4.1	1.6	2.9	1.9	to	2.3
REX3	-1.3	32.5	26.7	3.3	15.3	14.3	to	14.7
REX4	8.1	0.9	9.1	-0.5	4.4	3.4	to	3.8
REX5	2.8	2.3	4.0	1.3	2.6	1.6	to	2.0
REX6	3.5	2.8	3.6	2.8	3.2	2.2	to	2.6

Figure 64. Summary of adhesive bonds for the CH pads. All numbers have units of µm.

After bonding the last CH pad in place and allowing the adhesive to cure for 24 hours, the six pads were diamond turned to the proper thickness. A depth of cut of 1 μ m, a spindle speed of 1600 rpm, and a feedrate of 5 mm/min were used for the finishing passes. The LVDT verified

that the thickness of each of the ablators components, which consists of the CH material and the Stycast epoxy, was $32.8 \mu m$.

After machining the CH pads to thickness, the 100 mm Al disk was removed from the vacuum chuck. The face of the disk that had been against the vacuum chuck wad covered with a viscous substance. This substance could be removed with mineral spirits and a cotton swab.

The surface finish on the Cu appeared to have been altered by the vacuum chuck. Before placing the part into the vacuum chuck, the Cu had a lustrous finish, as shown in the picture on the left side of Figure 65. The small spots on the Cu in this image are residue from alcohol. After removing the part from the vacuum chuck, several features were visible on the 1 mm wide band of Cu in the locations of the radial grooves on the vacuum chuck. The center photograph in Figure 65 depicts the region on the Cu that had the most prominent features. The grooves of the vacuum chuck may have been mechanically or chemically imprinted into the Cu. In some isolated spots, the Cu contained pits with a depth of approximately 1 μ m. As shown in the picture on the right side of Figure 65, these features did not appear on the Cu regions between the radial grooves in the vacuum chuck.

The ICH and CH pads on disk D3 had been placed directly over the radial grooves in the vacuum chuck, so the Cu contained these features on the targets. However, the Cu on the targets still met the surface finish specification. In the future, the ICH and CH pads should NOT be bonded directly over the radial grooves of the vacuum chuck. In addition, the portion of the vacuum chuck that supports the Cu should be reconsidered. It may be preferable to eliminate the circumferential vacuum grooves and use a simple flat surface to prevent the Cu from deflecting. It might also be preferable to use a material in these vacuum grooves that will not cause chemical or mechanical damage to the Cu.



Figure 65. Freshly diamond turned 1 mm \times 43 μ m band of Cu (left). Cu that was adjacent to the radial grooves in the vacuum chuck (center). Cu that was not adjacent to the radial grooves in the vacuum chuck (right).

The center photograph in Figure 65 depicts the region of the Cu with the most prominent features. Most of the regions on the Cu that were in the vicinity of the radial grooves on the vacuum chuck contained features that were less prominent, as shown in Figure 66.



Figure 66. Features on the Cu in the vicinity of a radial groove on the vacuum chuck. The photograph is representative of most of the features on the Cu.

LASER MACHINING

The excimer laser in B131 was used to cut 4 mm diameter subassemblies from the 100 mm disk. The 4 mm pieces were cut out of the 6 mm diameter ICH and CH pads. To perform the laser machining, the 100 mm disk was placed on a brass surface. The side of the disk containing the Cu was placed against the brass, and the laser spot was incident from the top onto the CH pads. The initial cuts were made in part ICH1-REX4, which had a cracked CH pad. Images of this part appear in Figure 67.



Figure 67. Part ICH1-REX4. CH side of the disk (left) and Al-Cu side of the disk (right).

Several observations were made from this trial. The CH and/or the ICH adjacent to the edge of the 4 mm part appears to have been burned. When machining parts with the excimer laser, a sheet of paper is typically placed between the part and the surface supporting the part. However, in this case, paper was not used between the brass chuck and the workpiece. Without any paper beneath the part, the laser may have reflected off of the brass and could have created the white features near the edge of the aluminum, which are apparent in the right side of Figure 67. The zinc in the brass chuck has a low vapor pressure and may have been ablated by the laser. Because of the profile of the workpiece and the shape of the brass chuck, after the laser had finished machining, the 4 mm part fell through the newly created hole. As it fell, the laser may have hit it and created the blackened spot that is apparent in the left side of Figure 67. Note that on the opposite face of the part, there is a raised feature at the location of the blackened spot.

Several trials were performed to develop a laser machining process that eliminated each of these problems. Testing revealed that an Ar gas shield actually caused the plastic layers to blacken even further. Therefore, an Ar gas shield should not be used when laser machining these

parts. To reduce the burn marks on the part, the laser machining parameters were adjusted. The parameters that were used on subsequent parts are listed in Figure 68.

Parameters	Value	Parameters	Value
System Operator:	Brian Kelly	Travel Speed/VV	0.04
Data File Name: (example)	EOS2nd	Vector Acceleration	N/A
Wavelength:	248	Kerf Width	~0.050mm
FF Lens	LMU-5X-248	# of Passes Through	one
Focus Distance	1.424"	Shutter Dwell (seconds)	0.2
Entrance Aperture (pin hole)	.285" Cu	Displayed Values:	
Image Aperture Size (mask)	0.100"	HV (KV)	14
Demag	24.4:1	Power (W)	65
Energy, mJ	360	Energy, (mJ)	300
Rep Rate, Hz	180		

Figure 68. Parameters used for the excimer laser

To eliminate any burn marks at the location where the laser initially enters the workpiece, the laser should enter the workpiece several mm from the location of the part and should tangentially move onto the diameter of the part. To reduce the back reflection from the brass chuck, one or more pieces of paper should be placed between the workpiece and the chuck. A technique was also developed to prevent the part from falling through the newly machined hole at the conclusion of the cut. Rather than cutting the full 360° of the 4 mm circle, only 356° was cut, leaving a 4° span of material. This tab of material held the 4 mm disk in place until the laser was turned off. The part could then be held with a vacuum pencil and gently rocked back and forth to break the tab. This process left only a small burr on the edge of the 4 mm part.

The laser cutting process left a white residue on the surface of the Al. Most of this residue could be removed using a dry cotton swab. After removing most of the residue, the parts could be cleaned further using a cotton swab containing alcohol. However, it is best to **not use** any alcohol on the parts until most of the debris has been removed, because the alcohol can actually causes debris to bond to the surface of the aluminum.

MEASUREMENT OF SURFACE FINISH

Each part was cleaned by removing as much debris as possible with a dry cotton swab and then wiping it with a cotton swab containing alcohol. The surface finish of the Cu and Al in the area of interest on the target were measured on the Wyko NT 8000 optical profiling system using PSI mode at 20×. The measurements for target ICH6-REX1 are shown below. After removing the tilt and cylindrical form from the dataset for the Cu surface, Sa=10 nm and St=192 nm, as shown in Figure 69. A profile trace through the dataset for the Cu had Ra=12 nm and Rt=108 nm, as shown in Figure 70. Therefore, the Cu met the specification of Ra<25 nm.

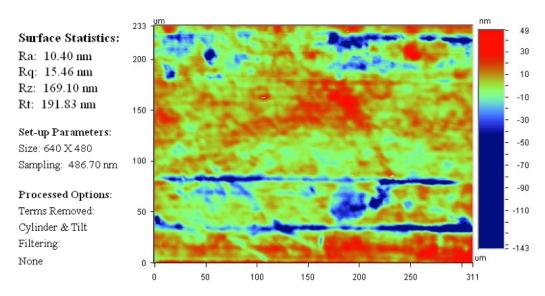


Figure 69. Wyko PSI 20× measurement of the Cu in the area of interest of ICH6-REX1. Tilt and cylindrical form have been removed from the dataset.

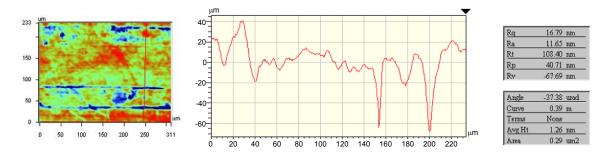


Figure 70. Profile trace through the Wyko dataset for the Cu of ICH6-REX1.

After removing the tilt and cylindrical form from the dataset for the Al surface, Sa=15 nm and St=167 nm, as shown in Figure 71. A profile trace through the dataset for the Al had Ra=13 nm and Rt=63 nm, as shown in Figure 72. Therefore, the Al met the specification of Ra<25 nm.

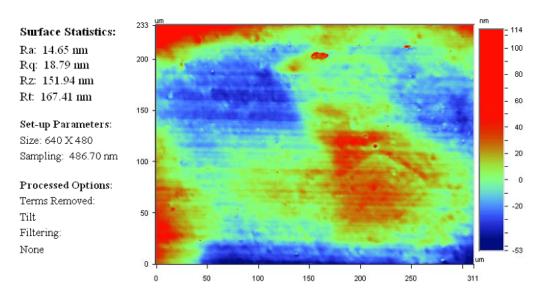


Figure 71. Wyko PSI 20× measurement of the Al in the area of interest of ICH6-REX1. Tilt and cylindrical form have been removed from the dataset.

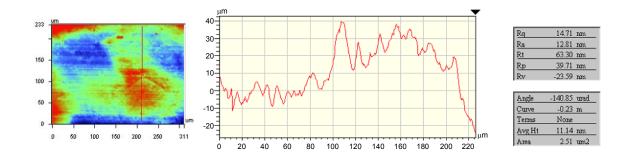


Figure 72. Profile trace through the Wyko dataset for the Al of ICH6-REX1.

BONDING WINDOWS TO THE Cu SAMPLE AND AI BASEPLATE

The thickness of several 1 mm \times 1.4 mm quartz windows was measured by placing each window on a vacuum chuck and measuring the step height from the surface of the window to the surface of the chuck using the Wyko NT 8000 optical profiling system in VSI mode at $2.5\times$. Each of the windows used on the targets had a measured thickness of $102~\mu m$. The uncertainty of this measurement was approximately 1 μm . Windows were bonded onto the Cu sample and the Al baseplate of the target using the benchtop air-bearing assembly station. The 4 mm subassembly consisting of the Cu, Al, ICH, and CH was placed onto a vacuum chuck in the lower actuator of the assembly station. Using a plastic vacuum pipette located in the upper actuator of the assembly station, each window was carefully aligned into position, as shown in Figure 73. A droplet of LocTite 3104 UV curing adhesive was used to bond the window in place. After pressing the window into the target with a force of 0.01 N, the UV adhesive was cured with the handheld Norland UV light.



Figure 73. Alignment of a quartz window on an EOS target

Windows were bonded to several test pieces and to several targets. Figure 74 contains microscope images of windows bonded to three different targets.

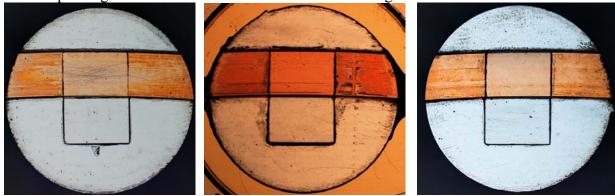


Figure 74. Windows bonded to three EOS targets

After bonding each window in place, the thickness of the adhesive bond was determined by measuring the step height from the surface of the window to the substrate using the Wyko NT 8000 optical profiling system in VSI mode with a Modulation Threshold of 2%. Of the ten windows bonded in this effort, eight of them met the specification of an adhesive bond of less than 5 μ m. The adhesive bonds that did not meet the specification probably resulted from debris on the windows that prevented them from seating correctly on the substrates. It is very important that the windows be cleaned as well as possible before being bonded, and the assembly area should be as clean as possible to prevent debris from interfering with assembly.

Figure 75 contains a plot of the Wyko measurement of the quartz window on the Al baseplate of target ICH6-REX1. This measurement was made after bonding the window to the Al, but before a window was bonded to the Cu. The numbers in the yellow boxes in the figure represent the measured step heights from the surface of the window to the Al baseplate at various locations around the window. The window had a thickness of $102\pm1~\mu m$, so the thickness of the adhesive bond in the area of interest of the target is approximately $1\pm1~\mu m$.

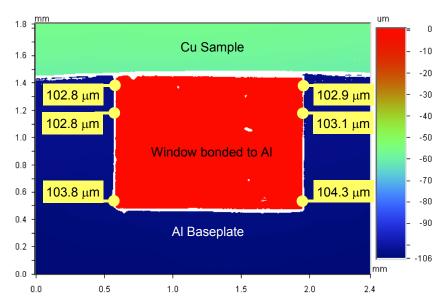


Figure 75. Wyko measurement of the window on the Al baseplate on target ICH6-REX1

Figure 76 contains a plot of the Wyko measurement of the quartz window on the Cu sample of target ICH6-REX1. This measurement was made after bonding the window to the Cu. The numbers in the yellow boxes in the figure represent the measured step heights from the surface of the window to the Cu substrate at various locations around the window. The window had a thickness of $102\pm1~\mu m$, so the thickness of the adhesive bond in the area of interest of the target is approximately $2\pm1~\mu m$. Note that outside the area of interest, the adhesive reached a thickness of approximately $5\pm1~\mu m$.

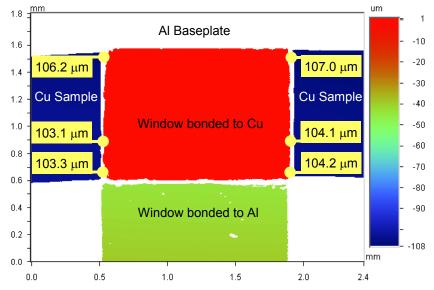


Figure 76. Wyko measurement of the window on the Cu sample on target ICH6-REX1

COMPLETION OF A Cu EOS TARGET

Target ICH6-REX1 was completed by bonding the target to a gold support ring. Figure 77 depicts the target just before it was bonded to the support ring.

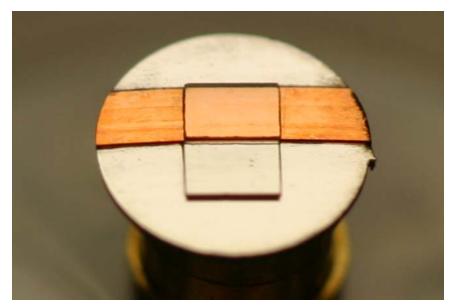


Figure 77. Target ICH6-REX1 after bonding the windows but before bonding the support ring
Figure 78 depicts the completed Cu EOS target after bonding it to a gold support ring. This

target met all of the fabrication and metrology specifications.



Figure 78. Complete Cu EOS target

CONCLUSION

The Cu Equation of State laser target was successfully fabricated and met all of the manufacturing and metrology specifications. The target consisted of planar layers of copper, aluminum, iodine-doped polystyrene, and pure polystyrene. The required thickness uniformities of the various layers of the target ranged from $\pm 0.1~\mu m$ to $\pm 0.3~\mu m$, and they had to be measured with this same level of accuracy. To meet the requirements for the interface between the aluminum and the copper, a sputter-seeded electroplating process was developed to deposit copper onto the aluminum. The thickness uniformity requirements for the various layers were obtained by fabricating the targets on a 100 mm diameter aluminum disk that was held on the

diamond turning machine using carefully designed vacuum chucks. By performing assembly and metrology directly on the diamond turning machine, the workpiece could remain in the vacuum chuck throughout the manufacturing process. Therefore, the reference surfaces required for the fabrication and metrology of the targets could be maintained, and the targets could be manufactured to meet all of the specifications.

RECOMMENDATIONS

There are a number of ways in which the process developed in this effort could be improved. The chuck that held the 100 mm Al disk on the diamond turning machine while it was being machined to a thickness of 62 μm created features on the Cu sample. The chuck contained a recess of width 1 mm and depth 43 μm to accommodate the band of Cu on the 100 mm Al disk. The bottom surface of this recess contained several circumferential vacuum grooves that were connected to the other grooves on the chuck by eight radial grooves. In the vicinity of these radial grooves, the form of the vacuum grooves became imprinted into the Cu. Fortunately, even with these features imprinted in the Cu, the targets met the surface finish specification. However, in the future, steps should be taken to prevent any damage occurring to the Cu. It may be possible to eliminate the circumferential vacuum grooves and use a simple flat surface on the bottom of the 1 mm \times 43 μm recess in the chuck. It might also be preferable to use a material in these vacuum grooves that will not cause chemical or mechanical damage to the Cu. Furthermore, the ICH and CH pads should not be bonded directly over the radial grooves of the vacuum chuck.

After the 100 mm Al disk was machined to a thickness of 62 μ m and was removed from the vacuum chuck, the surface was covered with a viscous substance. This substance may have been cutting fluid that had been pulled beneath the workpiece by the vacuum. To prevent any fluid from penetrating beneath the workpiece, its perimeter could be bonded to the chuck using urethane or viscous vacuum grease.

One of the biggest obstacles to achieving thin, uniform layers of adhesive between the components of the target was debris on the surfaces. Chips generated in the machining processes, dust, and debris from containers used to store components had a tendency to stick to the ICH pads, CH pads, and quartz windows. It was extremely important to clean all parts carefully before assembling them. All parts must be stored in clean containers that do not generate particles. For example, glass containers that have been carefully cleaned work well. Plastic bags should never be used to store components. To eliminate the possibility of dust settling onto components, assembly must be performed in a clean room. All parts must be carefully examined with a microscope to ensure their cleanliness.

Great care must be exercised when using alcohol to clean parts. In many cases, alcohol can be used to clean parts thoroughly. However, alcohol can dissolve certain materials and weld particles to the surface, making it even more difficult to remove them. Therefore, before using alcohol to remove particles from a surface, it must first be tested to determine its effectiveness.

When fabricating the ICH and CH pads that formed the pre-heat shield and ablator, the diamond parting tool used to machine the back surface was not long enough to reach the center of the 6 mm diameter pads. Therefore, a HSS parting tool was used to machine the center section of the back surface, which created a 4 mm rough area on each pad. When measuring the thickness of the pads using the LVDT, the rough area had to be cantilevered over a hole in the chuck, which caused the pads to deflect and led to an error of up to 1 µm in the measured thickness. To avoid this problem, the back surface of each pad must be machined using a

diamond parting tool that is long enough to reach the center of the pad. If the entire back surface of the pad is diamond turned, then the entire back surface of the pad could be supported by a vacuum chuck while its thickness is being measured.

StyCast epoxy was used to bond the CH pads on top of the ICH pre-heat shields. This epoxy must cure for approximately 8 hours before the assembly cone can be removed from the CH pad. Therefore, only a single CH pad could be bonded each day. Hardman Green epoxy could be used in place of the StyCast to bond the CH pads. Hardman Green epoxy requires only about 2 hours before the assembly cone can be removed from the CH pad, but it is slightly more viscous and produces bonds that are slightly thicker. The time required to apply the pre-heat shields and ablators could be reduced considerably if doped-epoxies were developed that met the material requirements for these components. The components could then be created by applying the epoxy, letting it cure, and then diamond turning it to the required thickness.

Future Equation of State targets may require a sample material than cannot be electroplated. For these targets, it may be necessary to sputter the sample material to the required thickness. Sputtering techniques must be developed to ensure that acceptable stresses are created in the sputtered material. Recall that in this study, the stresses created by sputtering 30 µm of Cu onto an Al substrate were not adequately studied. Therefore, it is not known if this process could have been used to fabricate targets.

ACKNOWLEDGEMENTS

This work would not have been possible without the skill and expertise of Carlos Castro, who performed the vast majority of the work in this effort. Many of the manufacturing and metrology details that made this effort a success originated with Carlos. His machining skills, knowledge of vacuum chuck design, and experience with precision metrology equipment made this work possible. The success of this effort is due largely to his hard work.

A number of other people also made contributions. Ron Foreman and Johann Lotscher contributed greatly to this project. They designed and executed the processes for sputtering the seed layer of Cu onto the Al. Thanks to their expertise, an acceptable interface between the Cu and the Al was achieved. Rudy Robles performed the electroplating, assisted with the identification of proper plating parameters, and helped design the hardmask used for electroplating. His efforts are also greatly appreciated. Steve Strodtbeck helped with some of the machining and assembly. Brian Kelly designed the laser machining process and performed all of the cutting with the excimer laser. Joe Satcher, Stuart Gammon, and Bob Reibold provided materials, identified proper solvents for removing cutting fluids from the parts, and provided general knowledge that was useful when designing the manufacturing process for the targets. Robin Miles applied the lithographic masks. Alan Jankowski performed some of the Cu deposition processes during the preliminary tests. Kuang Jen Wu performed SIMS analyses during the development of a cleaning process. Alex Hamza assisted with issues related to materials science. Marcia Kellam assisted with the quartz windows and measured the window thickness. Lee Griffith and Jeff Kass provided guidance and managerial support for this effort.

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Appendix A – Target Specifications

*thickness uniformity spec is written so that +/- means TOTAL wedge (from one side of piece to other) can be twice number quoted

*** Will measure 6 locations on the 100mm disk and interpolate to obtain values for the targets locations

[Numbers in brackets are for reference only; they are not acceptance criteria. The metrology of the feature is not critical to Physics.] It is understood btwn Physics & Target Fab that if the uncertainty in measuring the feature is comparable to the requested dimension of the feature, then the metrology will not prove if the desired mfg spec has been met.

Physics requirements for EOS targets Final Specs Set on 12/16/04						
Feature	Specification [microns]	Actual value [microns]	Method for verifying part is acceptable. Notes on acceptance criteria are indicated in blue			
Components	[IIIICIOIIS]	[IIIICIONS]	acceptance criteria are indicated in blue			
Baseplate (Al 1100)						
Nominal thickness of bare baseplate	62 +/- 2.0	63.0	Cap probe on DTM will measure from chuck face to			
Measurement uncertainty	1.0	< 0.2	baseplate face **			
Al step btwn sample & bare baseplate	< 1.0	0.02	LVDT on DTM will measure step **			
Measurement uncertainty	0.2	0.05	LVD1 On D1M will measure step			
Thickness uniformity of bare baseplate*	+/- 0.25	Satisfied	Cap probe on DTM will measure seating of 100mm disk on vacuum chuck **			
Acceptance Criterion	see note	0.15	The runout of the 100 mm diameter disk must be < 200 nm			
Thickness uniformity of Al under sample*	+/- 0.25	Satisfied	Cap probe on DTM will measure seating of 100mm disk on vacuum chuck **			
Acceptance Criterion	see note	0.15	Predicted max ∆ of 1mm wide band of sample matl with 1 atm on vac chuck is 94 nm. Acceptance criterion is: the runout of the 100 mm diameter disk must be < 200 nm with the sample side of the disk against the vacuum chuck.			
Surface roughness (Ra = average)	Ra < 25 nm on VISAR side Ra < 100 nm on	Ra = 13 nm Ra = 15 nm	Wyko, PSI mode - Measure directly on VISAR side ** Measure a witness area on the ICH side.			
Sample (Cu OEHC) for Cu targets only	ICH side	<u> </u>				
Sample (Cu OFHC) for Cu targets only	43+/- 2.0	42.0				
Nominal thickness		43.0	Wyko VSI mode **			
Measurement uncertainty	0.3	0.07	Cap probe on DTM will measure seating of 100mm disk			
Thickness uniformity*	+/- 0.1	Satisfied	on vacuum chuck **			
Acceptance Criterion	see note	0.20	The runout of the 100 mm diameter disk must be < 200 mm when machining Cu			
Surface roughness (Ra = average)	Ra < 25 nm	Ra = 12 nm	Wyko, PSI mode. Can only measure window side **			
Bulk density, % TMD	> 96%, if homogeneous	Satisfied	Rutherford Back Scattering (RBS)			
Measurement uncertainty, %	0.5	Satisfied				
Preheat Shield (6.25 at% I-CH)+adhesive						
Nominal thickness (ICH+adhesive)	102 +/- 3	103.6	LVDT on DTM will measure 4 points on machined ICH, 4			
Measurement uncertainty	2.5	0.05	pts on baseplate			
Thickness uniformity (ICH+adhesive)*	[+/- 0.25]	< 0.05	LVDT on DTM will measure 4 points on machined ICH, 4			
Measurement uncertainty	[0.25]	0.05	pts on baseplate			
Surface roughness (peak-valley)	Rt < 600 nm	Rt = 80 nm	Measure one witness sample with Wyko PSI mode			
Ablator (CH)+adhesive						
Nominal thickness (CH+adhesive)	32.5 +/- 2.5	32.8	LVDT on DTM will measure 4 points on surface of final			
Measurement uncertainty	[2.5]	0.07	machined CH, 4 pts on baseplate			
Thickness uniformity of (glue+ICH+glue+CH) subassembly *	[+/- 0.3]	< 0.07	LVDT on DTM will measure 4 points on surface of final			
Measurement uncertainty	[0.3]	0.07	machined CH, 4 pts on baseplate			
Surface roughness (peak-valley)	Rt < 600 nm	Rt = 80 nm	Wyko, PSI mode. Will measure one part only.			
Window (alpha quartz)						
Nominal thickness	100 +/- 25	102	ATMM, SEM, measuring mic, Wyko, or laser micrometer			
Measurement uncertainty	1.0	1	(Target Fab will choose method)			
Thickness uniformity*	[+/- 0.25]	< 1	ATMM, SEM, measuring mic, Wyko, or laser micrometer (Target Fab will choose method)			
Acceptance Criterion	see note		Target Fab will choose the method and will provide an estimate of the uncertainty of the measurement. No method has yet been proven to have an uncertainty of 0.25 µm or better.			
Flatness	+/- 0.25	< 0.12	Wyko, PSI mode			
A.R. coating for reflective 0.5-1.5% at	659 nm		4 <u>1 </u>			
Al coating thickness, angstroms	0	1				

Appendix A – Target Specifications

	Specification	Actual value	Method for verifying part is acceptable. Notes on
Feature	[microns]	[microns]	acceptance criteria are indicated in blue
Assembly			
Baseplate-Sample joint			
Thickness	< 0.5	< 0.1	Cu will be deposited onto the Al baseplate with a 20
Thickness uniformity*	+/- 0.1	< 0.1	nm adhesion layer of Ti. There is no need to
Measurement uncertainty	0.1	< 0.1	metrologize the interface.
Baseplate-Preheat Shield joint	•	•	
Thickness	[< 3.0]	≤ 2.80	Measure ICH pad before bonding, measure "ICH+adhesive" with LVDT after bonding. Metrology
Thickness uniformity*	[+/- 1.5]	≤ 1.40	data will consist of ICH thickness at 4 pts with uncertainty of ±1-2 µm, and "ICH+adhesive" thickness at 4 pts with uncertainty of ±0.1 µm. Acceptance
Measurement uncertainty	[3.0] see note	+0.05 -1.0	criterion is: (nominal measured "ICH+adhesive" thickness) - (nominal measured ICH thickness) < 5 µm
Preheat Shield-Ablator joint			
Thickness	[< 3.0]	≤ 2.36	Measure CH pad before bonding, measure "CH+adhesive" with LVDT after bonding. Metrology
Thickness uniformity*	[+/- 1.5]	≤ 1.18	data will consist of CH thickness at 4 pts with uncertainty of ±1-2 µm, and "CH+adhesive" thickness at 4 pts with uncertainty of ±0.1 µm. Acceptance
Measurement uncertainty	[3.0] see note	+0.05 -1.0	criterion is: (nominal measured "CH+adhesive" thickness) - (nominal measured CH thickness) < 5 μm
Subassembly (w/o windows)			
Thickness uniformity*	+/- 0.3	Satisfied	LVDT on DTM will measure 4 points on surface of final machined CH, 4 pts on baseplate
Acceptance Criterion	see note	0.15	The runout of the 100 mm diameter disk must be < 200 nm
Flatness	20	< 3	Wyko, VSI mode
Baseplate-Wndw & Sample-Wndw joint			
Thickness	[< 5.0]	2	Measure "window+adhesive" with Wyko VSI after bonding. Metrology data will consist of
Thickness uniformity*	[+/- 1.5]	1	aforementioned window metrology, and "window+adhesive" thickness at 4 pts with uncertainty of ±0.2 µm. Acceptance criterion is: (nominal
Measurement uncertainty	[1.5]	< 1.5	measured "window+adhesive" thickness) - (nominal measured window thickness) < 6 μm

Process Sheet for Diamond Turned 1" Al Disk - Sample #3

		Performed	
Date	Task	by	Notes
8/20/04	Diamond turn face using mineral spirits	Strodtbeck	
8/20/04	Measure surface finish with Wyko	Bono	Rt=30 nm at 20x
8/20/04	Clean with pentane, acetone, isopropyl (5 times)	Bono	Last rinse used a 0.2um syringe filter with ethanol
8/20/04	Apply masking paint and Kapton tape	Bono	Applied a stripe of Bob Dively Liquid Masking Film (paint) through the center of the part. Paint dried for 2.5 hours
8/20/04	Peel off Kapton and paint	Bono	
8/20/04	Apply additional Kapton over part of the surface	Bono	There is a small smudge on the Al adjacent to the Kapton tape. This may be a fingerprint.
8/20/04	Clean surface with alcohol to try to remove the smudge on the surface of the Al	Lotscher	Smudge was removed.
8/20/04	SEM images of part Minimal cleaning prior to sputtering. This may	Sedillo	
	include oakite, DI water, and/or alcohol; these have all been verified to have negligble effect on surface finish. (Do not use anything that will attack Kapton or its adhesive- DO NOT use		
8/23/04	acetone)	Lotscher	Cleaned with ethanol only, followed by Ar blow dry
8/23/04	Ion mill surface prior to sputtering	Lotscher	Ion mill for 30 seconds, location=100 deg
8/23/04	Sputter 5 microns of Cu onto surface. You may use an adhesion layer of 20 nm of Ti.	Lotscher	Sputtered just over 5 microns of Cu.
	Electroplate additional Cu	Robles	
9/1/04	Diamond turn Cu to ~40 micron thickness	Castro	Insufficient Cu was deposited. Cu was machined to about 30 micron thickness, but it did not clean up entirely. There were no problems with adhesion. Cu could not be pried up with a razor blade.

FINISHED

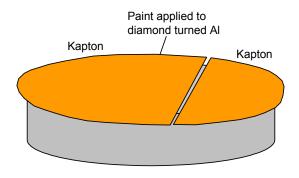
We later discovered on 9/14/04 that ethanol created a white debris when applied to a masked 4" disk with 1mm of exposed Al.

The following process was used to examine Sample #3.

Examining residue left by paint, and adhesion of sputtered Cu to masked diamond turned Al 1100

Please contact Matt Bono 2-5883 with any questions.

The objectives are 1) to determine how much residue is left after peeling off masking paint 2) to determine how well sputtered Cu with an adhesion layer of 10-20 nm of Ti adheres to a diamond turned disk of Al 1100 after it has been masked with paint/Kapton

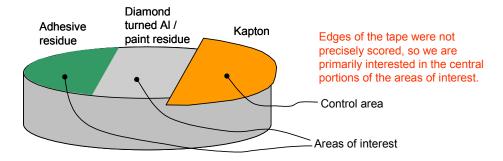


The sample has been diamond turned to Rt=30-40 nm, Rq=3-7 nm.

Kapton tape has been placed in two locations on the diamond turned surface.

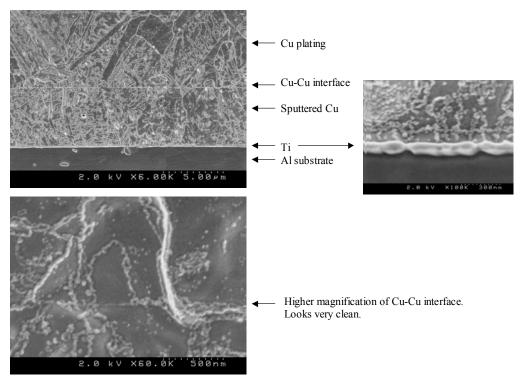
(The sample contains some isolated scratches, and we know their size and locations.)

- Peel off <u>the larger</u> piece of Kapton, taking the paint off in the process. Leave the other piece of Kapton in place. Be careful not to scratch or otherwise damage the diamond turned surface.
- Examine the central portions of the two areas of interest with an SEM to determine how
 much residue was left by the paint and by the Kapton adhesive. Please save these SEM
 images for us.

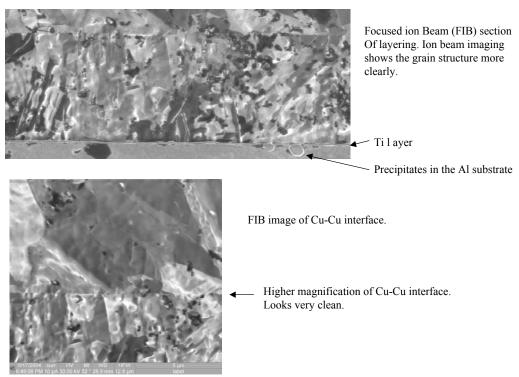


- Perform the normal cleaning procedure used for sputtering (it may include: aqueous detergent wash, DI and/or alcohol rinse, pre-sputter, ion milling, sputtering of an adhesion layer of Ti, etc.) Use the procedure you would normally use to get good adhesion when sputtering Cu onto AI in the presence of Kapton tape.
- Repeat the SEM measurements on each of the areas of interest and save the images
- Sputter 5 µm of Cu onto the disk
- Return the sputtered sample and SEM images to Matt Bono (2-5883).
- Please also provide details of the process used to clean / ion mill / pre-sputter the sample.

8/17/2004



SEM images of polished Sample #3



SEM images of FIB machined Sample #3

Process Sheet for Diamond Turned 1" Al Disk - Sample #4

		Performed	
Date	Task	by	Notes
8/19/04	Diamond turn face using mineral spirits	Castro	
8/19/04	Measure surface finish with Wyko	Bono	Rt = 30 nm at 20x, 50 nm at 10x
8/19/04	Clean with pentane, acetone, isopropyl (3 times)	Bono	
8/19/04	Measure surface finish with Wyko	Bono	Cleaning did not affect surface finish
8/20/04	SIMS analysis of surface - focus on the inner portion, because the edges may be contaminated from Latex gloves	Kuang Jen Wu	Identified traces of hydrocarbons on the surface, which may result from outgassing of box used to store part. Overall, the surface looked fairly clean.
8/20/04	Measure surface finish with Wyko	Bono	Rt = 30 nm at 20x, SIMS does not appear to have affected surface finish
8/20/04	Apply masking paint and Kapton tape	Bono	Applied a stripe of Bob Dively Liquid Masking Film (paint) through the center of the part
8/23/04	Peel off Kapton and paint	Bono	Held part using lens paper and nitrile clean room gloves while peeling paint
8/24/04	Alcohol rinse		
8/24/04	SIMS analysis of surface	Wu	Did not see any peaks representing the signature of either Mineral Spirits or Masking Paint.
Х	Reclean parts, if necessary	Bono	
9/2/04	Apply additional Kapton over part of the surface	Bono	Put two pieces of Kapton on the surface, leaving a stripe of exposed Al through the center of the part
Х	Measure surface finish with Wyko	Bono	
9/13/04	Minimal cleaning prior to sputtering. Use only ethanol. (Do not use anything that will attack Kapton or its adhesive- DO NOT use acetone, DO NOT use methanol)	Lotscher	Part was blown with Ar gas
9/13/04	Ion mill surface prior to sputtering	Lotscher	Part was ion milled for 30 seconds. Part was then stored in a Fluoroware container and taken to Kuang Jen Wu on 9/14/04. There was no evidence of Kapton redeposited on the
9/17/04 ?	SIMS analysis of surface	Wu	surface. However, there were several new substances that could have been residue from the tape that was redeposited on the surface.
Х	Sputter 5 microns of Cu onto surface	Foreman	
Х	Electroplate additional Cu	Robles	
Х	Diamond turn Cu to ~43 micron thickness	Castro	

FINISHED

The samples we run did not show additional Kapton deposit on the A1" Al sample. The SIMS spectra are fairly comparable to those run previously in the low mass region (m/z<100).

However, there are quite a number of additional peaks showed up at the mass range between ~150 - 500, the most pronounced peak is at m/z=149. Several significant peaks at high mass region suggest phthalate-related compounds. There might be other compounds present at the surface as well, albeit none of those peaks matched with any of the known compounds listed in our TOF-SIMS database. Judging by the intensities of these high mass peaks, these compounds are in relatively low surface concentration.

It is possible that these phthalate-related compounds were released from the tapes, which used to covered the portion disk surface.

We later discovered on 9/14/04 that ethanol created a white debris when applied to a masked 4" disk with 1mm of exposed Al.

Process Sheet for Diamond Turned 1" Al Disk - Sample #5

		Performed	Nata
Date	Task	by	Notes
8/19/04	Diamond turn face using mineral spirits	Castro	
8/19/04	Measure surface finish with Wyko	Bono	Rt = 30 nm at 20x, 50 nm at 10x
8/19/04	Clean with acetone and isopropyl (3 times)	Bono	The first of the 3 rinses included pentane. The last two did not.
8/19/04	Measure surface finish with Wyko	Bono	Cleaning did not affect surface finish
8/20/04	SIMS analysis of surface - focus on the inner portion, because the edges may be contaminated from Latex gloves	Kuang Jen Wu	Identified traces of hydrocarbons on the surface, which may result from outgassing of box used to store part. Overall, the surface looked fairly clean.
8/20/04	Measure surface finish with Wyko	Bono	Rt = 30 nm at 20x, SIMS does not appear to have affected surface finish
8/20/04	Apply masking paint and Kapton tape	Bono	Applied a stripe of Bob Dively Liquid Masking Film (paint) through the center of the part
8/23/04	Peel off Kapton and paint	Bono	Held part using lens paper and nitrile clean room gloves while peeling paint
	Alcohol rinse	Wu	
9/3/04	SIMS analysis of surface	Wu	Did not see any peaks representing the signature of either Mineral Spirits or Masking Paint.
STOP Finished	Apply additional Kapton over part of the surface	Bono	The remaining steps have already been performed on Sample 3
х	Measure surface finish with Wyko	Bono	
x	Minimal cleaning prior to sputtering. Use only ethanol. (Do not use anything that will attack Kapton or its adhesive- DO NOT use acetone, DO NOT use methanol)	Foreman	
X	ton mill surface prior to sputtering. This sample should NOT be ion milled	. oroman	
^x	Sputter 5 microns of Cu onto surface	Foreman	
X	Electroplate additional Cu	Robles	
X	Diamond turn Cu to ~43 micron thickness	Castro	
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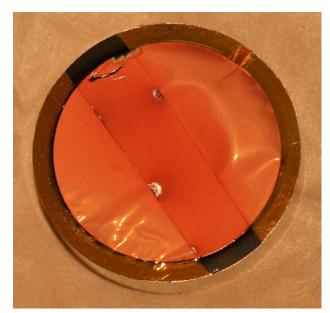
We later discovered on 9/14/04 that ethanol created a white debris when applied to a masked 4" disk with 1mm of exposed Al.

Process Sheet for Diamond Turned 1" Al Disk - Sample #1c

		Performed	
Date	Task	by	Notes
8/27/04	Diamond turn face using mineral spirits	Castro	Removed at least 50 microns of material. Part was drag-wiped after machining.
8/31/04	Measure surface finish with Wyko	Bono	Rt = 30 nm at 20x PSI
8/31/04	Clean with pentane, acetone, ethanol (3 times) Store Pentane in glass container.	Bono	Pentane was squirted onto part using a glass eyedropper pipette
8/31/04	Clean with oakite and rinse with DI water and ethanol	Bono	Oakite swirl for 30 seconds in a glass dish. Rinsed 6 times with DI water. Rinsed with ethanol.
8/31/04	Measure surface finish with Wyko	Bono	Rt = 40 nm at 20x PSI. Cleaning with Oakite appears to have affected Rt by at most 10 nm.
8/31/04	Apply masking paint and Kapton tape	Bono	Applied a stripe of Bob Dively Liquid Masking Film (paint) through the center of the part
9/1/04	Peel off Kapton and paint	Bono	Held part using lens paper and nitrile clean room gloves while peeling paint
9/1/04	Apply additional Kapton over part of the surface	Bono	Put two pieces of Kapton on the surface, leaving a stripe of exposed Al through the center of the part
9/15/04	Minimal cleaning prior to E-Beam PVD. This may include an aqueous detergent and/or ETHANOL. (Do not use anything that will attack Kapton or its adhesive - DO NOT use acetone or Methanol)	Jankowski	Alan did not provide the deposition parameters.
9/15/04	Deposit 60-100 microns of Cu by E-Beam PVD	Jankowski	Deposited ~100 microns of Cu. NEED DEPOSITION PARAMETERS
9/16/04	Diamond turn Cu to ~43 micron thickness	Castro	The Cu near the warped Kapton could be flaked from the surface. Elsewhere on the disk, the Cu could not be pried up with a scalpel.

Finished

We discovered on 9/14/04 that ethanol created a white debris when applied to a masked 4" disk with 1mm of exposed Al.



Disk as received on 9/16/04 after E-Beam PVD

There are two spots where the Kapton has warped. Adjacent to these areas, the Cu does not adequately cover the AI, and there are patches of uncoated AI.



Images of Sample 1c

After scoring the Kapton with a razor blade and peeling it from the surface, much of the Kapton residue remained on the disk. In addition, there are burn marks beneath the Kapton in the area where the Kapton had warped, and there is a bit of Cu that made its way under the warped Kapton.

The adhesive residue was later washed off with acetone.

residue on surface

1" Diamond Turned Al Samples - Summary

Sample 4		Sample 5	Sample 3	Sample 1c
Diamond turn face using mineral spirits		Diamond turn face using mineral spirits	Diamond turn face using mineral spirits	Diamond turn face using mineral spirits
Measure surface finish with Wyko		Measure surface finish with Wyko	Measure surface finish with Wyko	Measure surface finish with Wyko
Clean with pentane, acetone, isopropyl (3 times)		Clean with acetone, isopropyl (NO PENTANE)	Clean with pentane, acetone, isopropyl (5 times)	Clean with pentane, acetone, isopropyl (3 times), Oakite
Measure surface finish with Wyko		Measure surface finish with Wyko	Measure surface finish with Wyko	Measure surface finish with Wyko
SIMS analysis of surface		SIMS analysis of surface		
Measure surface finish with Wyko		Measure surface finish with Wyko		
Apply masking paint and Kapton tape		Apply masking paint and Kapton tape	Apply masking paint and Kapton tape	Apply masking paint and Kapton tape
Peel off Kapton and paint		Peel off Kapton and paint	Peel off Kapton and paint	Peel off Kapton and paint
Alcohol rinse		Alcohol rinse		
SIMS analysis of surface		SIMS analysis of surface		
Reclean parts, if necessary	STOP	Reclean parts, if necessary		
Apply additional Kapton over part of the surface	х	Apply additional Kapton over part of the surface	Apply additional Kapton over part of the surface	Apply additional Kapton over part of the surface
Measure surface finish with Wyko	х	Measure surface finish with Wyko		
Minimal cleaning prior to sputtering. This may include ethanol	х	Minimal cleaning prior to sputtering. This may include ethanol	Minimal cleaning prior to sputtering. This may include ethanol	Minimal cleaning prior to sputtering. This may include oakite, DI water, and/or ethanol
lon mill surface prior to sputtering	х	DO NOT DO ANY ION MILLING	lon mill surface prior to sputtering	
SIMS analysis of surface	х	SIMS analysis of surface		
Sputter 5 microns of Cu onto surface	х	Sputter 5 microns of Cu onto surface	Sputter 5 microns of Cu onto surface	Deposit Cu using E- Beam PVD
Electroplate additional Cu	x	Electroplate additional Cu	Electroplate additional Cu	
Diamond turn Cu to ~43 micron thickness	x	Diamond turn Cu to ~43 micron thickness	Diamond turn Cu to ~43 micron thickness	Diamond turn Cu to ~43 micron thickness
Finished - No Kapton, but possibly adheisve				Finished - Cu adhered,

Finished - no issues

but the Kapton warped

Finished

Appendix C – Process for bonding ICH and CH pads for EOS targets

April 29, 2005

Bond an ICH pad to a polystyrene assembly cone using LocTite 3103
 UV curing adhesive. Apply the smallest possible amount to hold the
 ICH pad to the cone. This process is performed using a benchtop
 assembly station. The bead of adhesive must be very flexible, so the pad
 can tip-tilt on the cone to lie flat against the Al during assembly. It may
 be necessary to stretch out the bead of adhesive prior to curing it.



 Place the holder with the assembly cone and the ICH pad into the assembly station mounted on the tool holder of the DTM.
 Apply a small droplet of Hardman Green epoxy to the ICH pad using a hair. The epoxy must first be mixed with a plastic rod (do not use wood) and centrifuged to remove the bubbles generated from mixing the epoxy.



• Slowly bring the ICH pad into contact with the Al substrate. Press the pad into the Al until the force reaches 5 grams-force. Move the pad with respect to the Al in an X pattern, where the length of each of the four legs of the X is 0.75 mm. Press the pad harder into the Al until the force reaches 10 grams-force. Move the pad with respect to the Al in an X pattern, where the length of each of the four legs of the X is 0.75 mm.



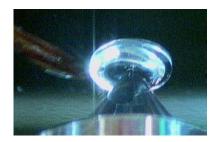
- After at least one hour, pull the assembly cone off of the ICH pad.
- Using an LVDT, measure the thickness of the resulting bondline.
- Allow the Hardman Green epoxy to cure for at least 24 hours.
- Machine the ICH pad to the final thickness. A finishing depth of cut of 5 μm works well for this machining step. If the depth of cut is too small, then the debris generated by the cutting process will stick to the surface of the ICH. Use a dry cotton swab to remove any debris from the surface of the ICH.



• Using an LVDT, measure the thickness of the resulting "adhesive-ICH pad" subassembly.

Appendix C – Process for bonding ICH and CH pads for EOS targets

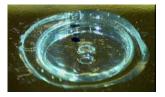
- Bond a CH pad to a polystyrene assembly cone using LocTite 3103 UV curing adhesive. Apply the smallest possible amount to hold the ICH pad to the cone. This process is performed using a benchtop assembly station.
- Place the holder with the assembly cone and the CH pad into the assembly station mounted on the tool holder of the DTM. Apply a small droplet of Stycast epoxy to the CH pad using a brush. A very small amount of epoxy should be applied; be careful not to apply too much. The epoxy must first be mixed with a plastic rod (do not use wood) and centrifuged to remove the bubbles generated from mixing.



• Slowly bring the CH pad into contact with the ICH. Press the pad into the ICH until the force reaches 5 grams-force. Move the pad with respect to the ICH in an X pattern, where the length of each of the four legs of the X is 0.5 mm. Press the pad harder into the ICH until the force reaches 10 grams-force. Move the pad with respect to the ICH in an X pattern, where the length of each of the four legs of the X is 0.5 mm.



• After at least one hour, pull the assembly cone off of the CH pad.



- Using an LVDT, measure the thickness of the resulting bondline.
- Allow the Stycast epoxy to cure for at least 2 days.
- Machine the CH pad to the final thickness. A finishing depth of cut of 5 µm works well for this machining step. If the depth of cut is too small, then the debris generated by the cutting process will stick to the surface of the CH. Use a dry cotton swab to remove any debris from the surface of the CH.
- Using an LVDT, measure the thickness of the resulting "adhesive-CH pad" subassembly.